



How Unstructured are Amorphous Polymer Melts? Solid-State NMR Studies of Local Dynamic Order in Amorphous Polymer Melts

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Introduction • interactions in solid state NMR

Solid State NMR • resolution enhancement in solid state NMR,
magic angle spinning, recoupling methods,
double quantum NMR spectroscopy.

I. Schnell, K. Saalwächter, M. Feike, R. Graf

Polymer dynamics • Reptation model, polybutadiene, PEMA

Conclusions • How unstructured are amorphous polymers ?



Molecular Structures and Dynamics via NMR



Important NMR interactions:

$$\mathbf{H} = \mathbf{H}_Z + \mathbf{H}_Q + \mathbf{H}_{CS} + \mathbf{H}_D + \mathbf{H}_J$$

Zeemann Interaction :

$$\mathbf{H}_Z = -\sum_i \gamma_i \underline{\mathbf{B}}_0 \underline{\mathbf{I}}^i$$

Quadrupol Interaction :

$$\mathbf{H}_Q = -\sum_i \frac{eQ}{2I(2I-1)\hbar} \underline{\mathbf{I}}^i \underline{\underline{\mathbf{V}}} \underline{\mathbf{I}}^i$$

Electronic Shielding :

$$\mathbf{H}_{CS} = -\sum_i \gamma_i \underline{\mathbf{B}}_0 \underline{\underline{\boldsymbol{\sigma}}} \underline{\mathbf{I}}^i$$

Dipol-Dipol Interaction :

$$\mathbf{H}_D = -\sum_{i \neq j} \frac{\mu_0 \hbar}{4\pi} \frac{\gamma_i \gamma_j}{r^3} \left[\frac{3}{r^2} (\underline{\mathbf{I}}^i \cdot \underline{\mathbf{r}}) (\underline{\mathbf{I}}^j \cdot \underline{\mathbf{r}}) - \underline{\mathbf{I}}^i \cdot \underline{\mathbf{I}}^j \right]$$

Indirect Spin-Spin Interaction : $\mathbf{H}_J = -\sum_{i \neq j} \underline{\mathbf{I}}^i \cdot \underline{\underline{\mathbf{J}}}^{ij} \underline{\mathbf{I}}^j$



Interactions in Solid State NMR Spectroscopy



Zeemann interaction dominates all other NMR interactions

Perturbation
Theory

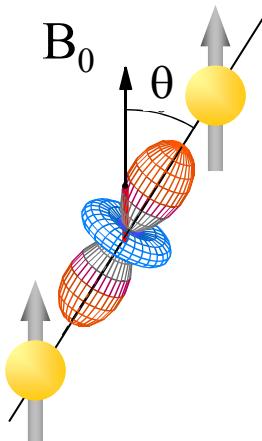
Orientation dependence of local spin interaction on B_0

Isotropic Contributions

H_{cs} : chemical shift

H_J : J-couplings

Liquid state NMR



H_Q, H_D, H_{cs}, H_J

Anisotropic Contributions

Symmetric

H_Q : quadrupol

H_D : dipol-dipol

H_{cs} : chemical shift

Asymmetric

H_Q : quadrupol

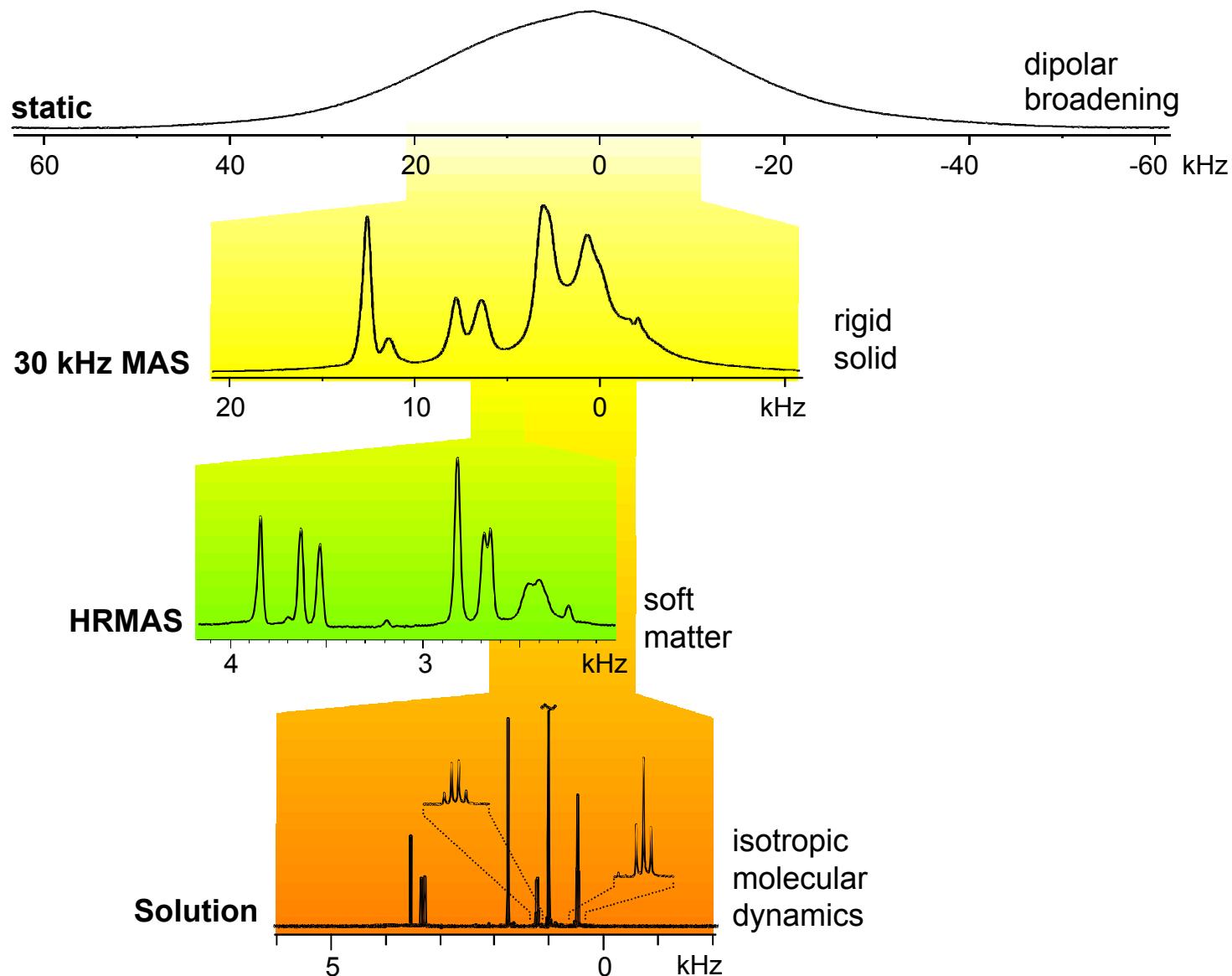
H_{cs} : chemical shift

$$H_D = \sum_{i \neq j} \frac{\mu_0 \hbar}{4\pi} \frac{\gamma_i \gamma_j}{r_{ij}^3} \frac{1}{2} (3 \cos^2 \theta_{ij} - 1) T_{2,0}^{ij}$$

Distance Orientation



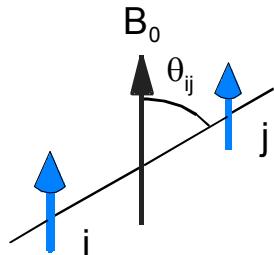
^1H NMR Spectra in Liquid and in Solid State



Spectral Resolution Enhancement in Solid State NMR



dipol-dipol coupling:



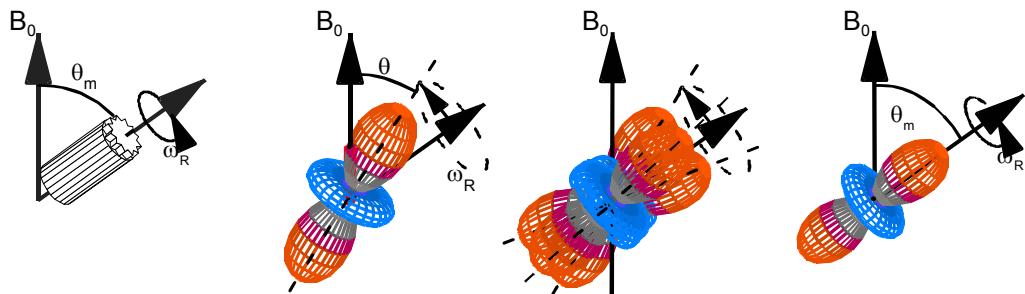
magic angle spinning:

$$\bar{\hat{R}}_{2,0} \rightarrow 0$$

$$\hat{H} = \hat{R}_{2,0} \cdot \hat{T}_{2,0}$$

$\hat{H} \propto \frac{1}{r_{ij}^3} \frac{1}{2} (3 \cos^2 \theta_{ij} - 1)$

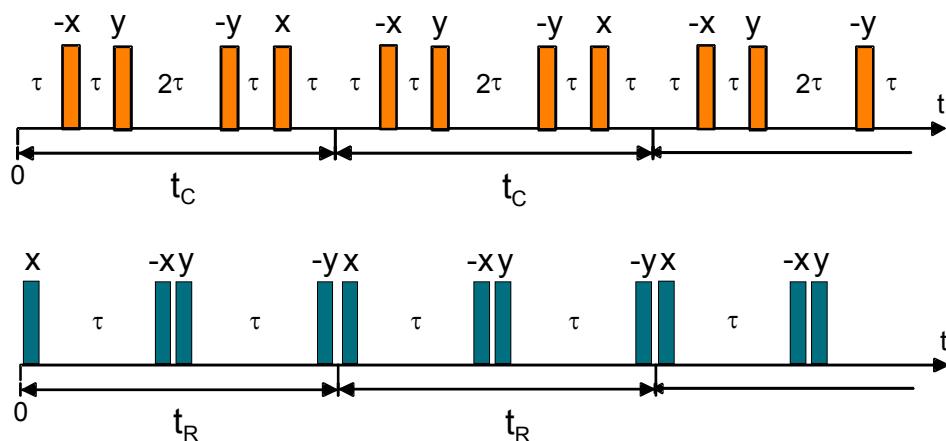
space
 $\gamma_i \gamma_j (3\hat{I}_{Z,i}\hat{I}_{Z,j} - \hat{I}_i \cdot \hat{I}_j)$



RF irradiation:

$$\begin{matrix} \bar{\hat{T}}_{2,0} & 0 \end{matrix} \quad (\text{CRAMPS})$$

$$\bar{\hat{T}}_{2,0} \quad H_{D,\text{eff.}} \quad (\text{Recoupling})$$



Double Quantum NMR Spectroscopy under MAS



preparation

evolution t_1

reconversion

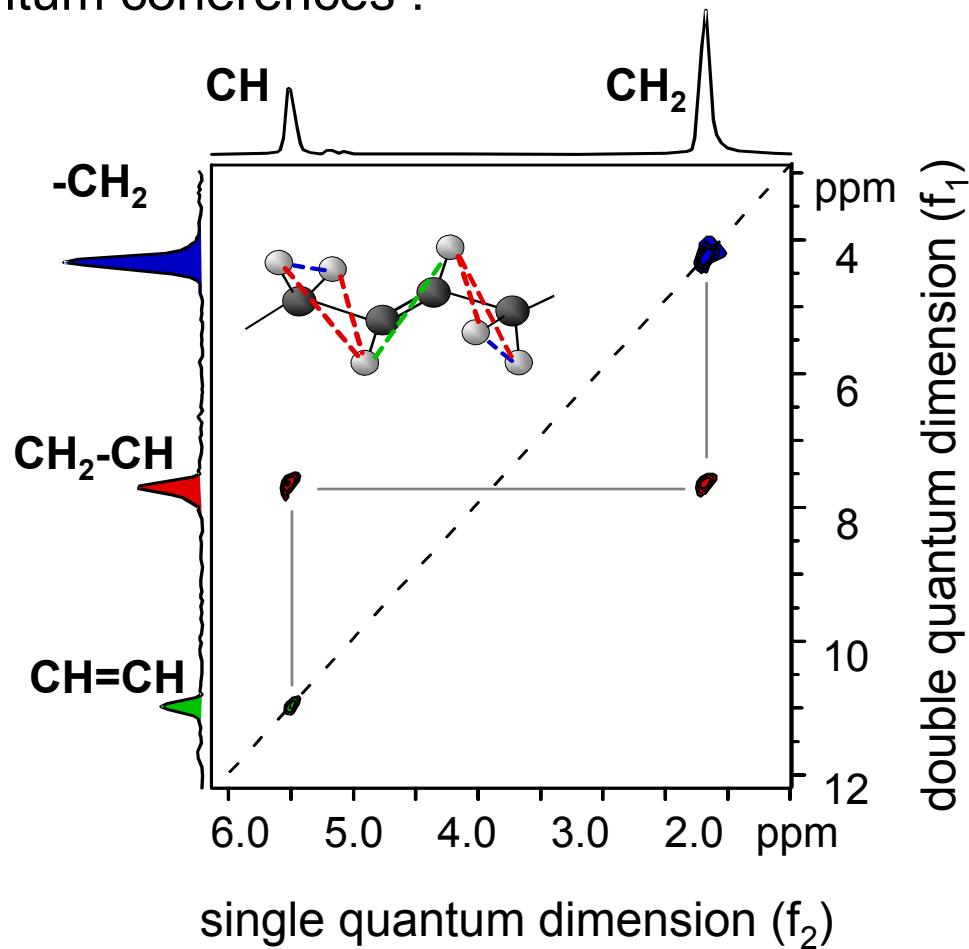
detection t_2

properties of double quantum coherences :

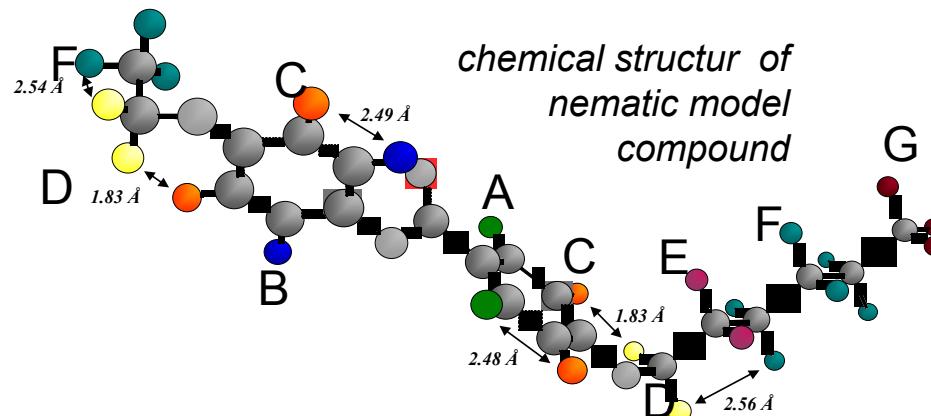
$$\omega_{DQ} = \sum_i \omega_{SQ,i}$$

$$I_{DQ,ij} = f(D_{ij} \cdot t)$$

$$\frac{dM}{dt} \approx 0$$



Order Parameter in Liquid Crystalline Phases

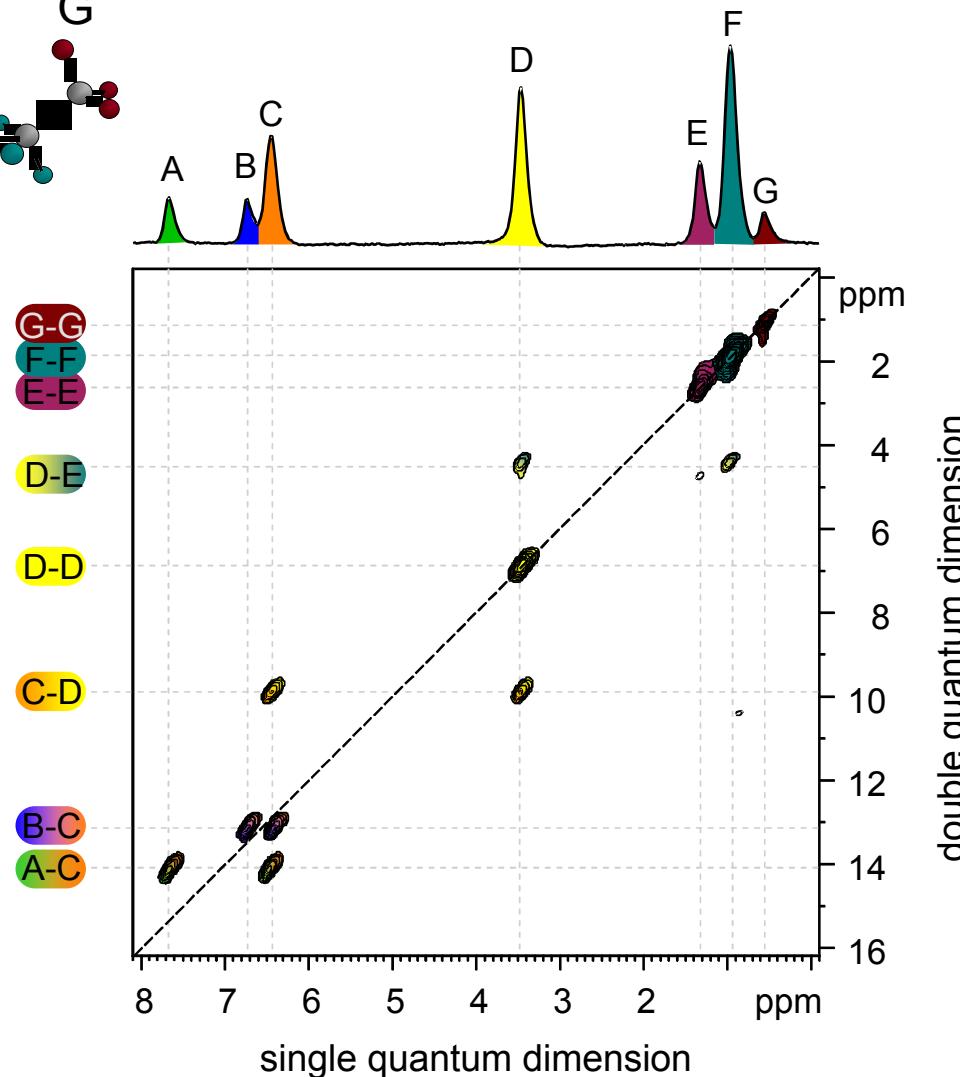


order parameter S_{ij} :

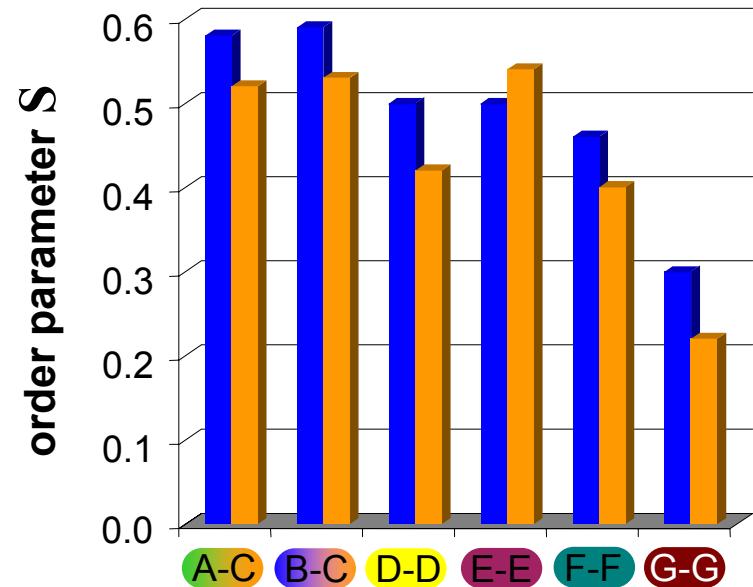
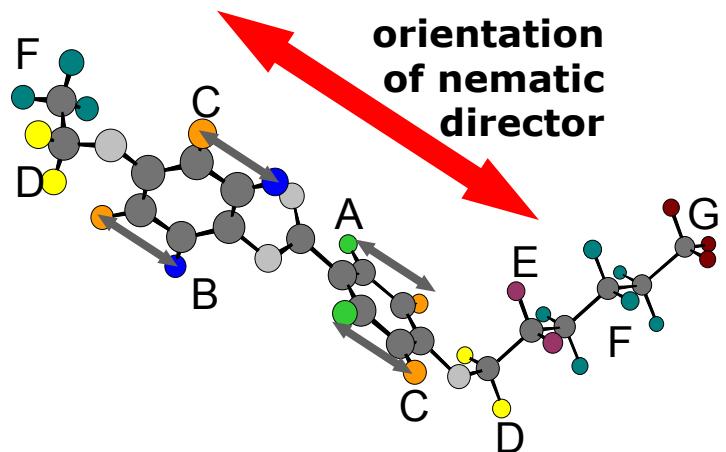
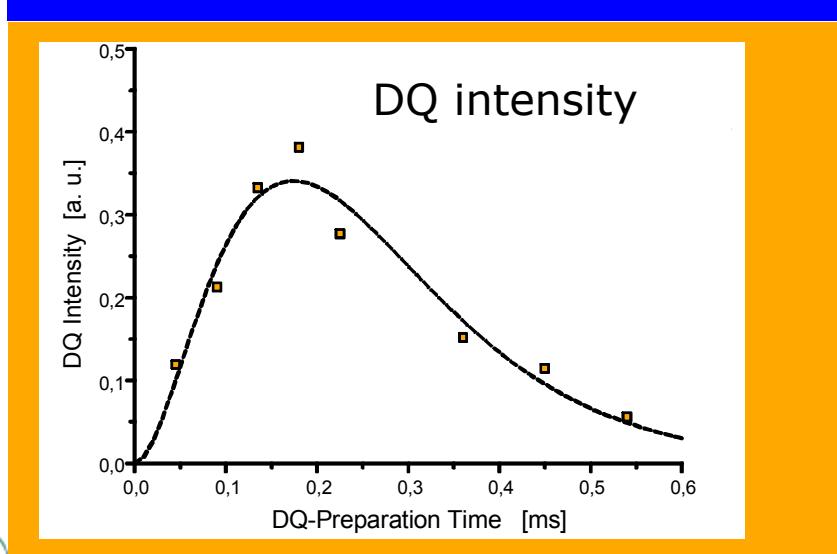
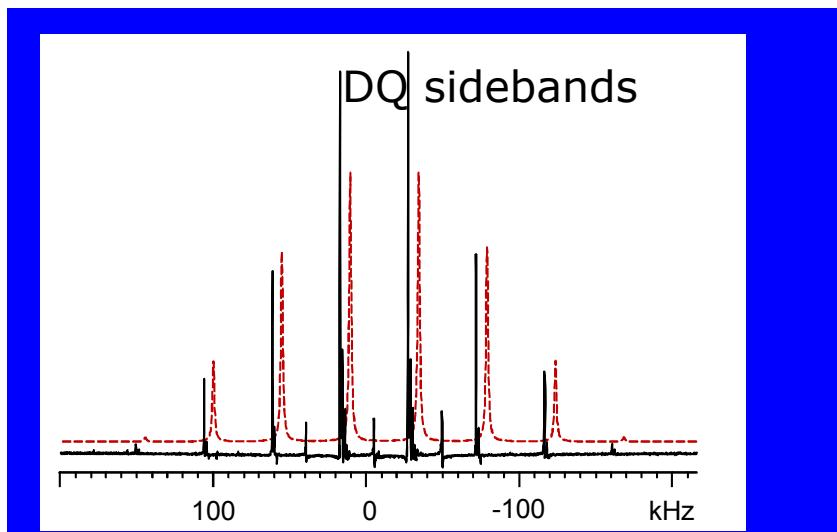
$$S_{ij} = \left\langle \frac{1}{2}(3 \cos^2 \theta - 1) \right\rangle$$



$$S_{ij} = \frac{\left\langle D_{ij,\text{eff}} \right\rangle}{D_{ij,\text{stat}}}$$



Order Parameters in Liquid Crystalline Systems

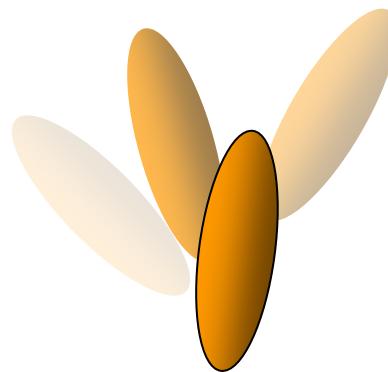


Local Order Parameter in Liquid Crystals and Polymers



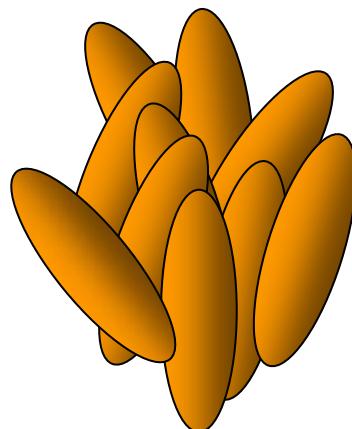
Nematic Systems

time

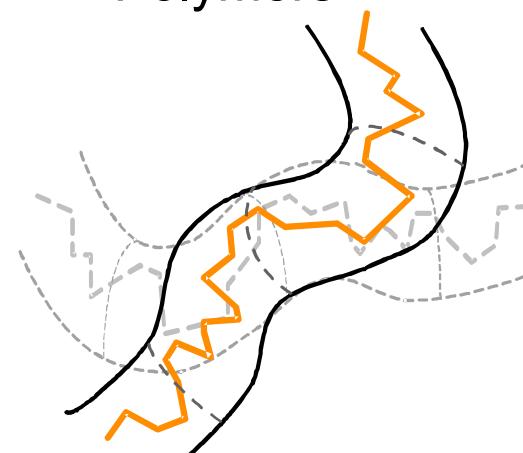


$$\langle S \rangle_t = \langle S \rangle_n$$

ensemble



Polymers



$$\langle S \rangle_t > 0, \langle S \rangle_N = 0, \\ \text{but } \langle S \rangle_{n \subseteq N} > 0$$





How Unstructured are Amorphous Polymer Melts? Solid-State NMR Studies of Local Dynamic Order in Amorphous Polymer Melts

Introduction • Interaction in solid state NMR

Solid State NMR • MAS, recoupling, double-quantum NMR

Polymer Dynamics • Reptations-model, scaling laws in polymer dynamics, influence of rigid confinements, conformational stability in PEMA melts.

T. Dollase, M. Neidhöfer, M. Wind, A. Heuer, R. Graf

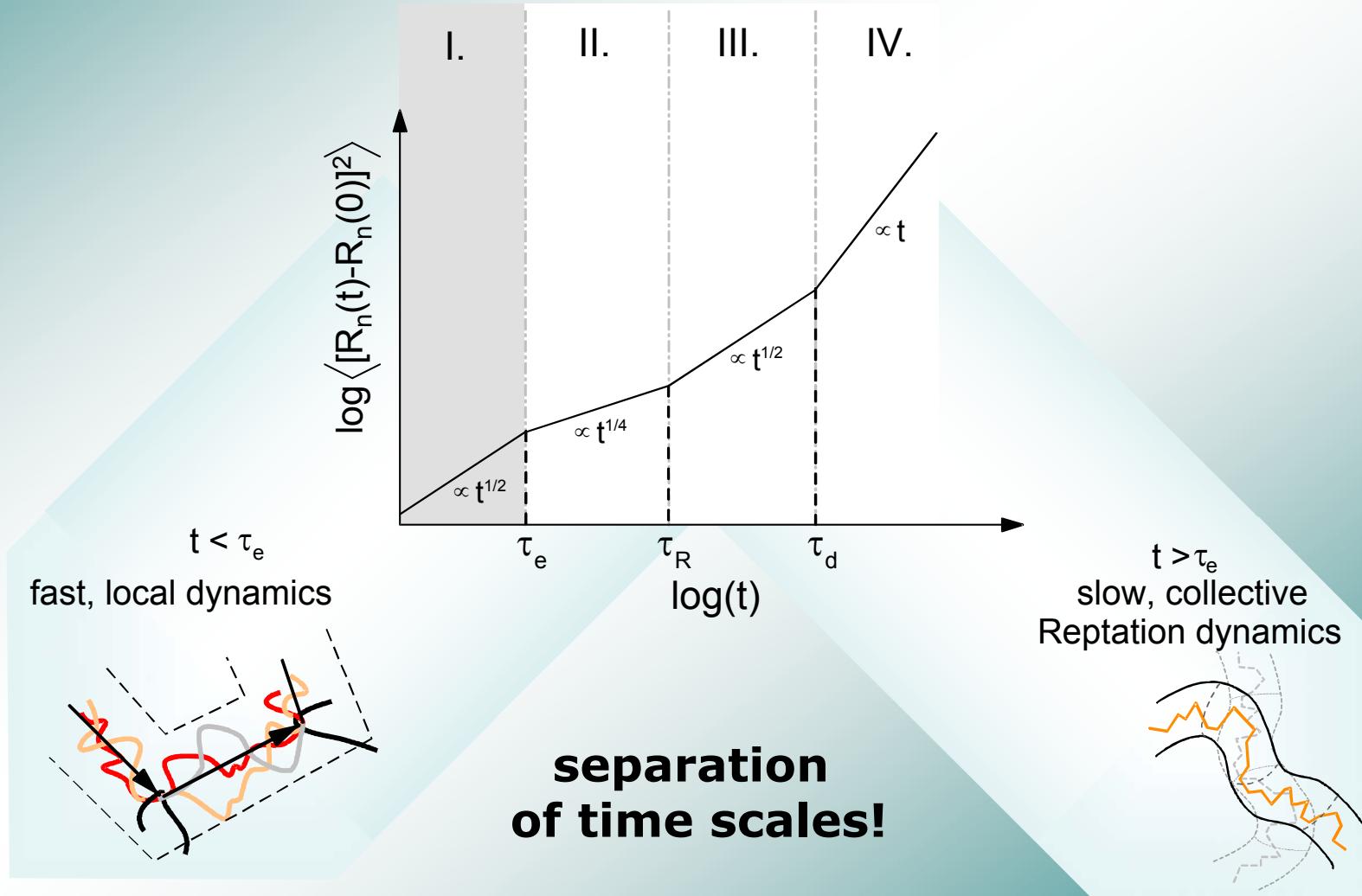
Conclusions • How unstructured are amorphous polymers ?



Length- and Time Scales in Polymer Dynamics



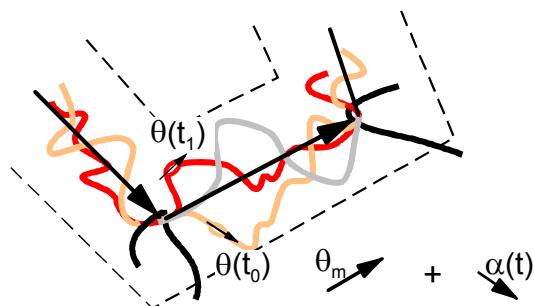
dynamic regimes of the Reptation model :



DQ Measurements of Dynamics on Different Time Scales



$$I_{DQ} \propto \left\langle \int_0^t dt' \int_{t+t_1}^{2t+t_1} dt'' \left\langle D_{ij,\text{eff}} \right\rangle^2 \cdot d_{2,-m}^{(2)}(t') d_{2,m}^{(2)}(t'') \right\rangle$$



local order parameter :

$$S_{ij} = \left\langle D_{ij,\text{eff}} \right\rangle / D_{ij}$$

static systems :

$$S_{ij}(t)=1$$

isotropic motion :

$$S_{ij}(t)=0$$

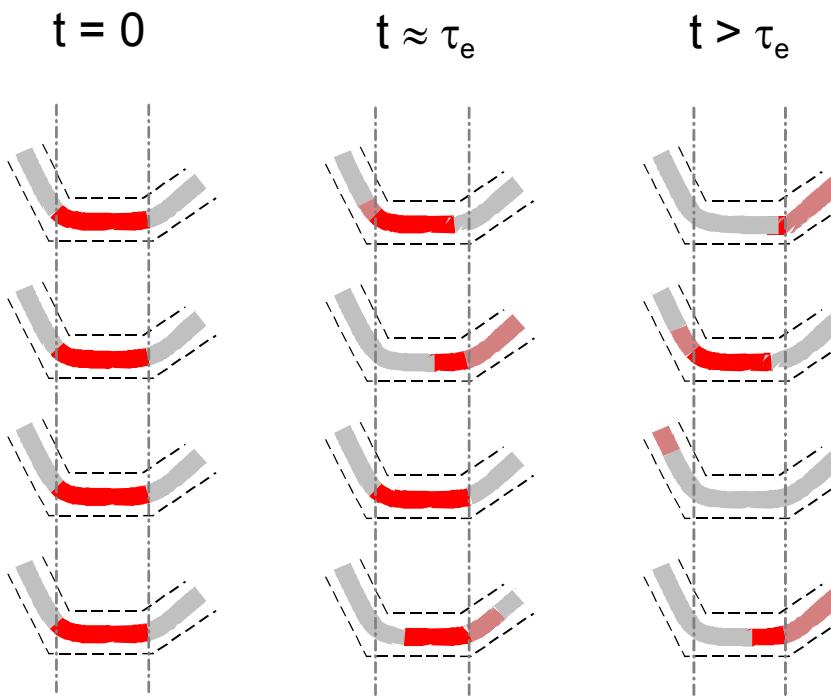
polymer network theory :

$$S \approx \frac{3}{5} N_e^{-1}$$



Polybutadien :

$$S \approx \frac{3}{5} \frac{M_{\text{Kuhn}}}{M_e} \approx 0.03$$

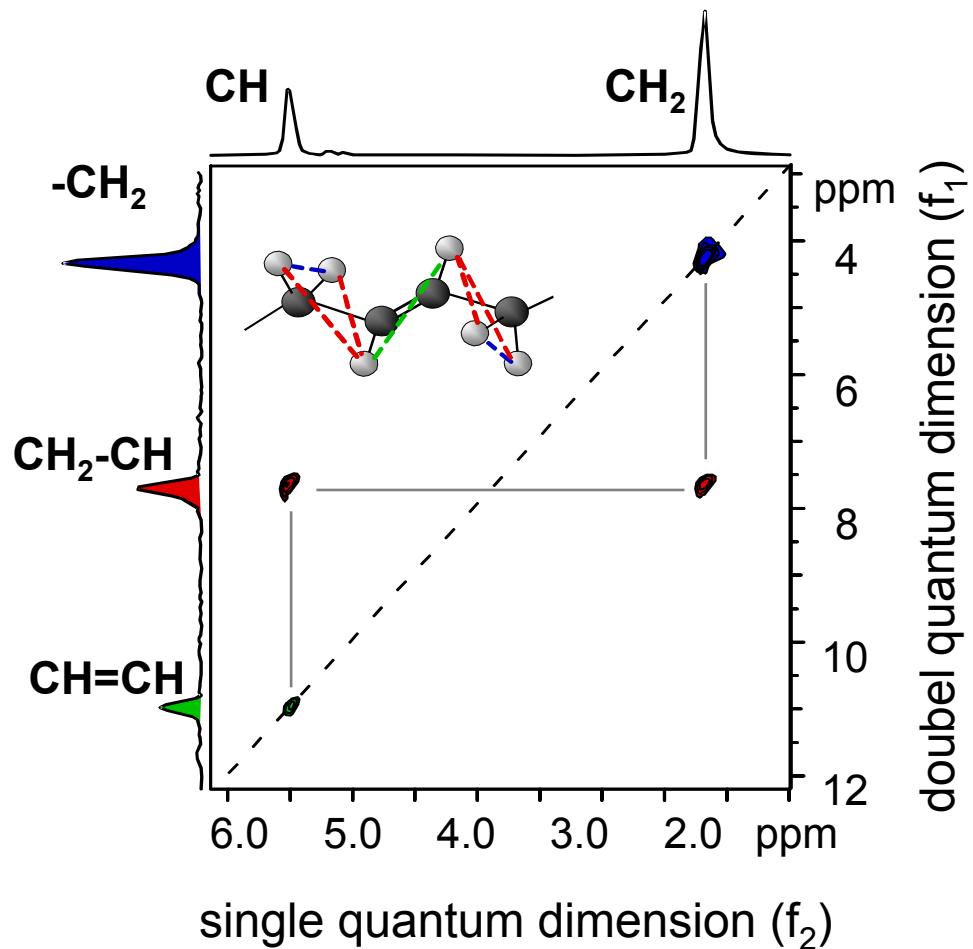


$\left\langle d_{2,-m}^{(2)}(t'_{\text{exc.}}) \cdot d_{2,m}^{(2)}(t''_{\text{rec.}}) \right\rangle_t$ corresponds to
return-to-origin probability $C(t)$

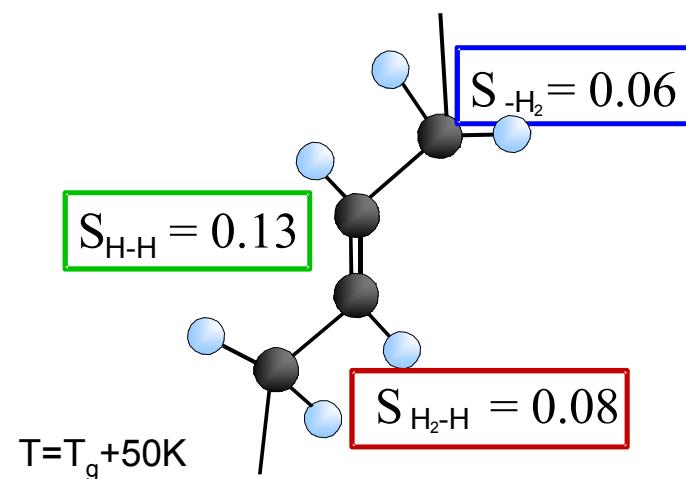
Local Order Parameters in 1,4 Polybutadien Melts



^1H double quantum NMR spectrum



Dynamic order parameter S
via residual dipolar couplings



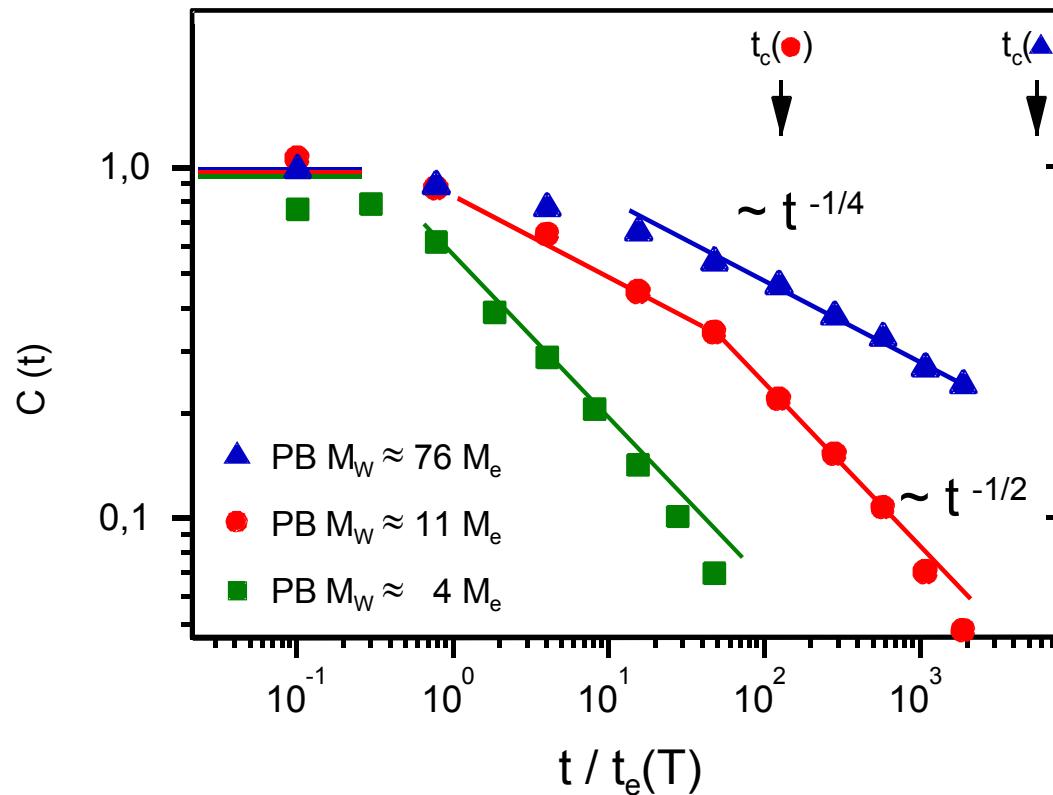
$$\Rightarrow S_{\text{C}=\text{C}} = 0.20 \pm 0.05$$



Time Dependence of Local Order Parameter



double-quantum filtered experiments on 1,4 poly-butadien

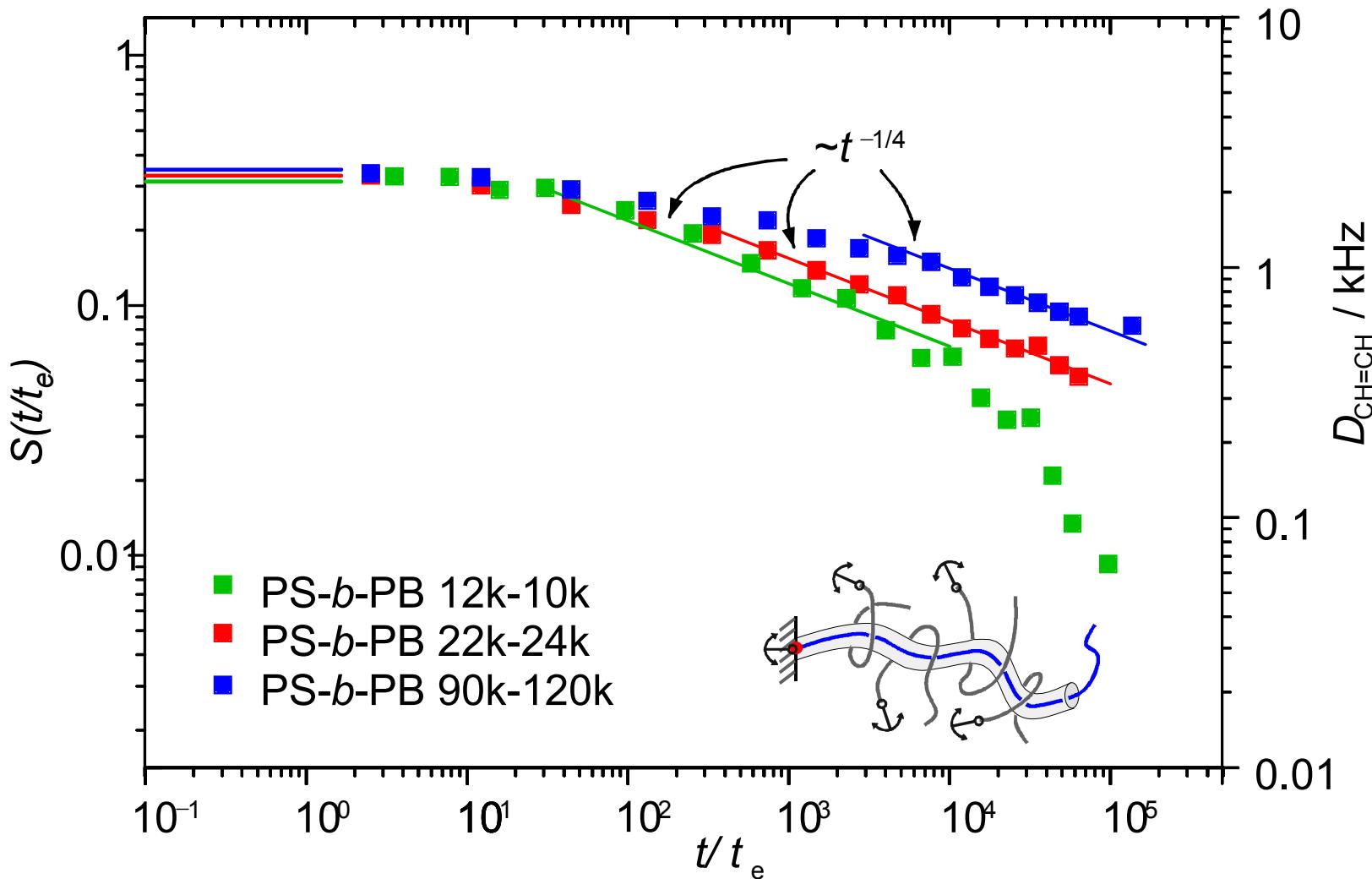


Reptation-model predicts two scaling laws:

$$S \sim t^{-1/4} \text{ and } S \sim t^{-1/2}$$

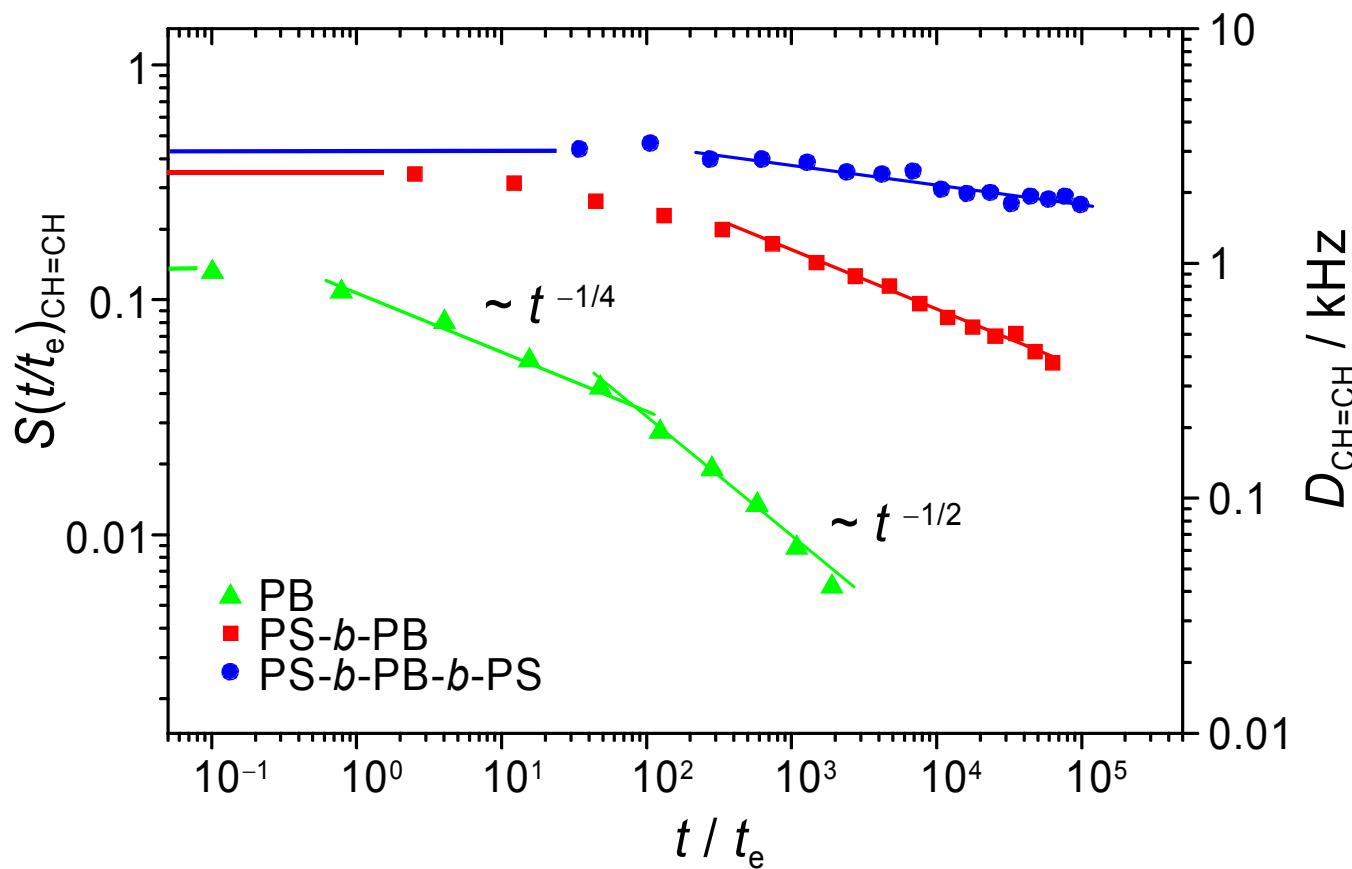


Molecular Weight Dependent Dynamics of PB Melts in PS-PB

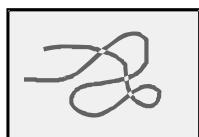
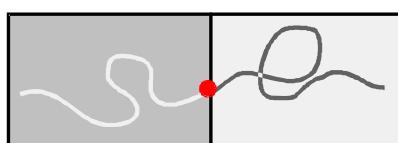


Tethering a PB chain end to a rigid PS block stabilizes the $t^{-1/4}$ -regime

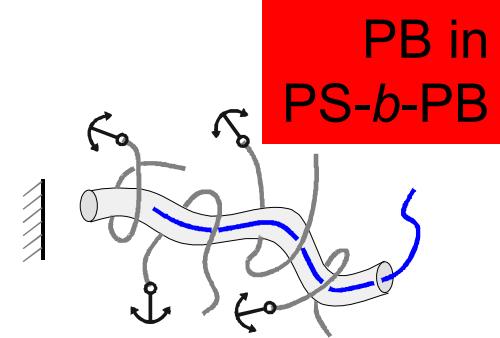
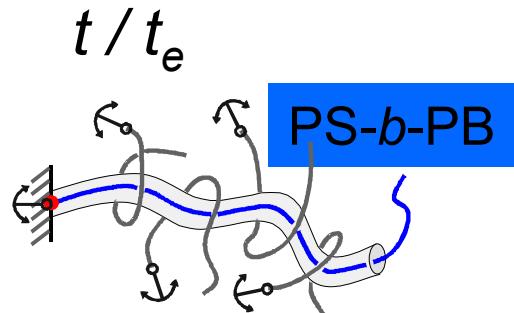
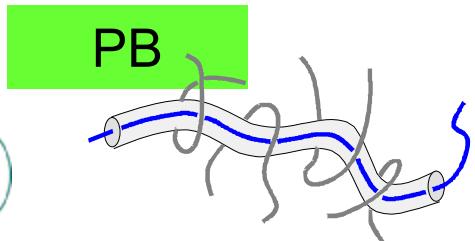
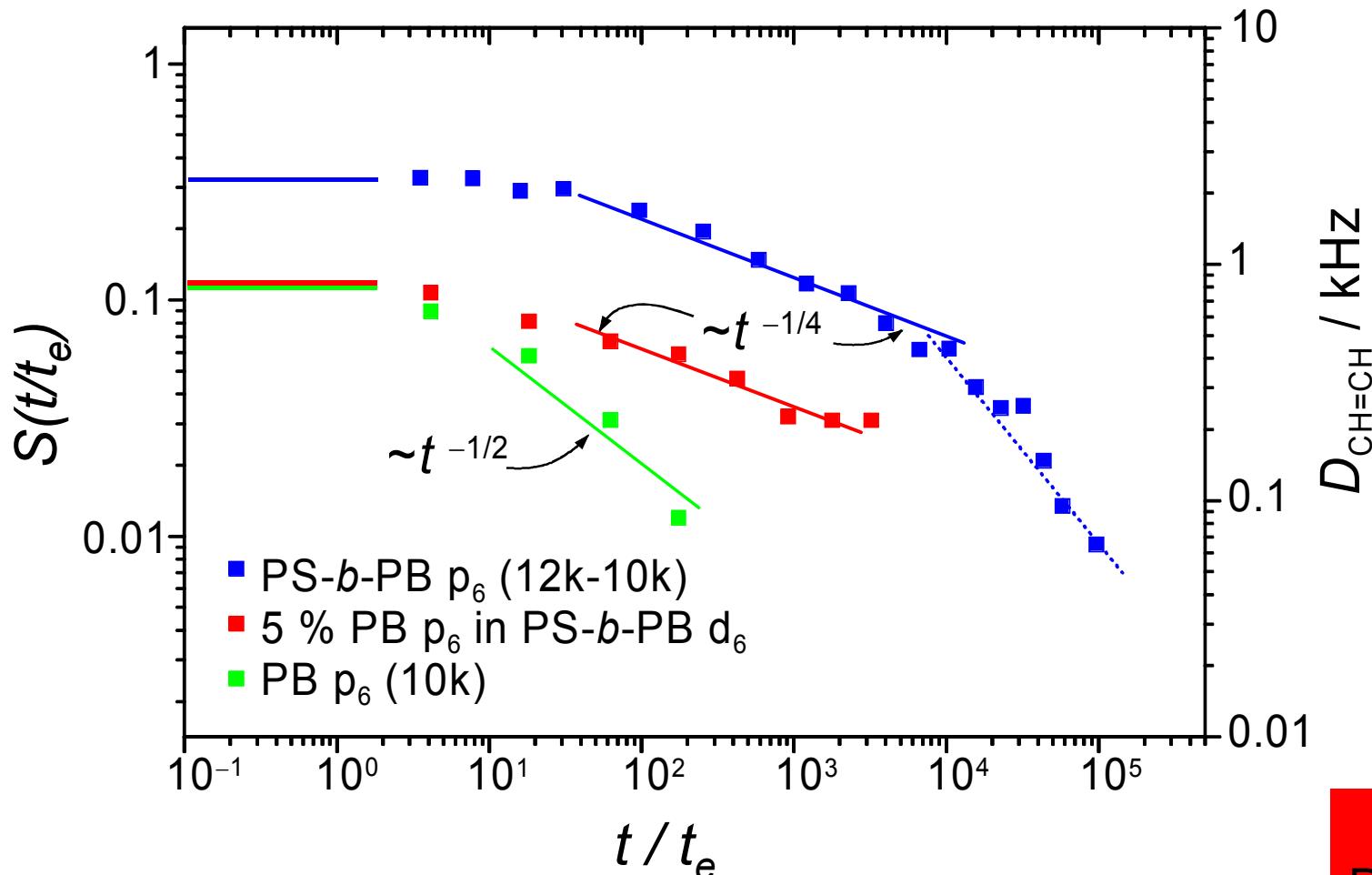
Influence of Rigid Confinements on Polymer Dynamics



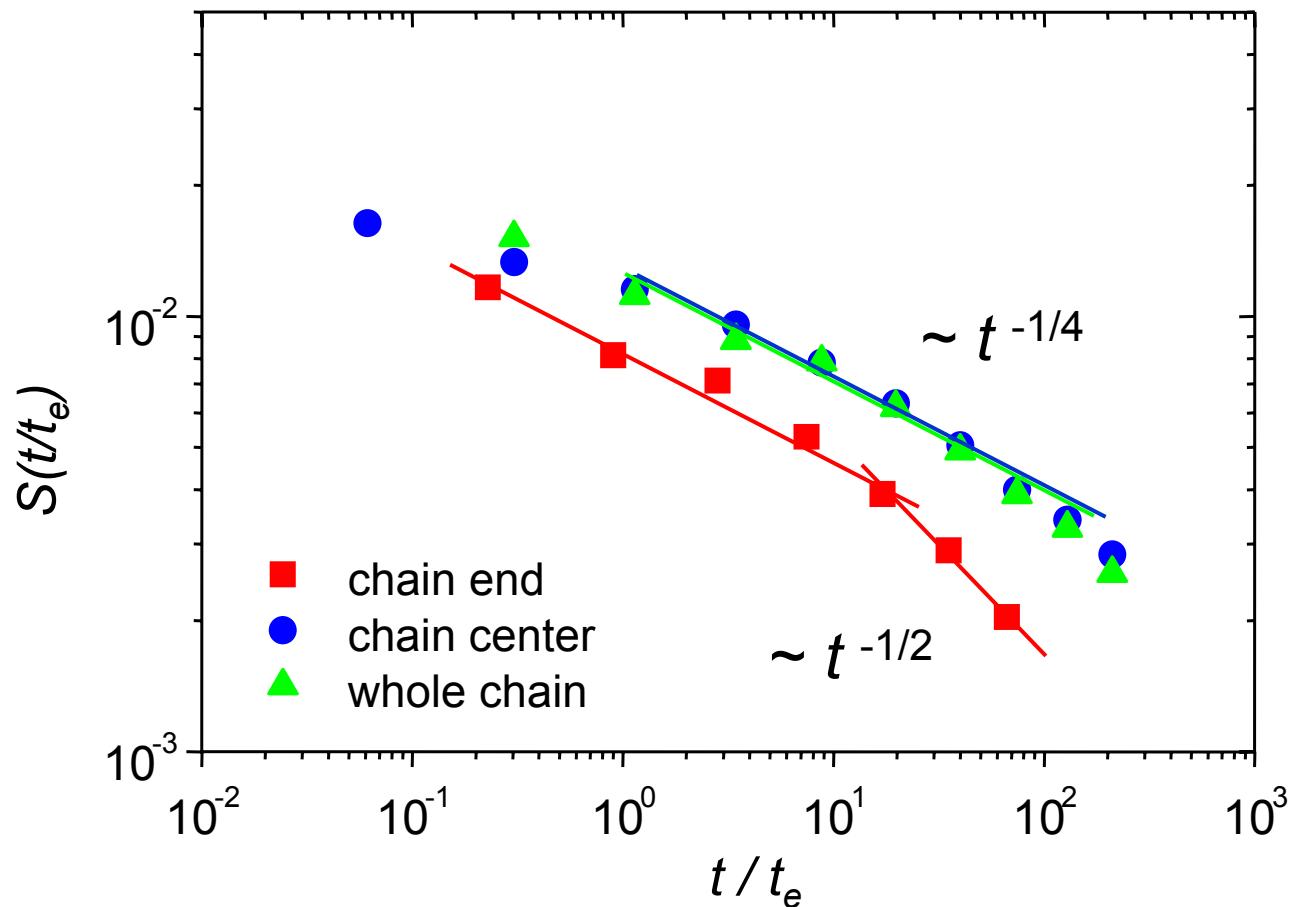
PB

PS-*b*-PBPS-*b*-PB-*b*-PS

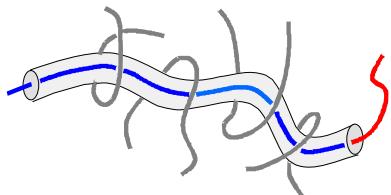
Polymer Dynamics in heterogeneous Polymer Melts



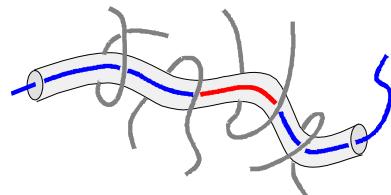
Variation of Dynamic Order Along the Polymer Chain



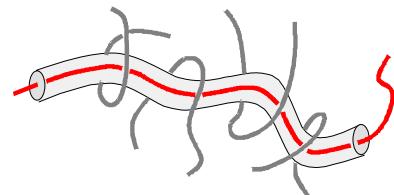
PB(d_6)-PB



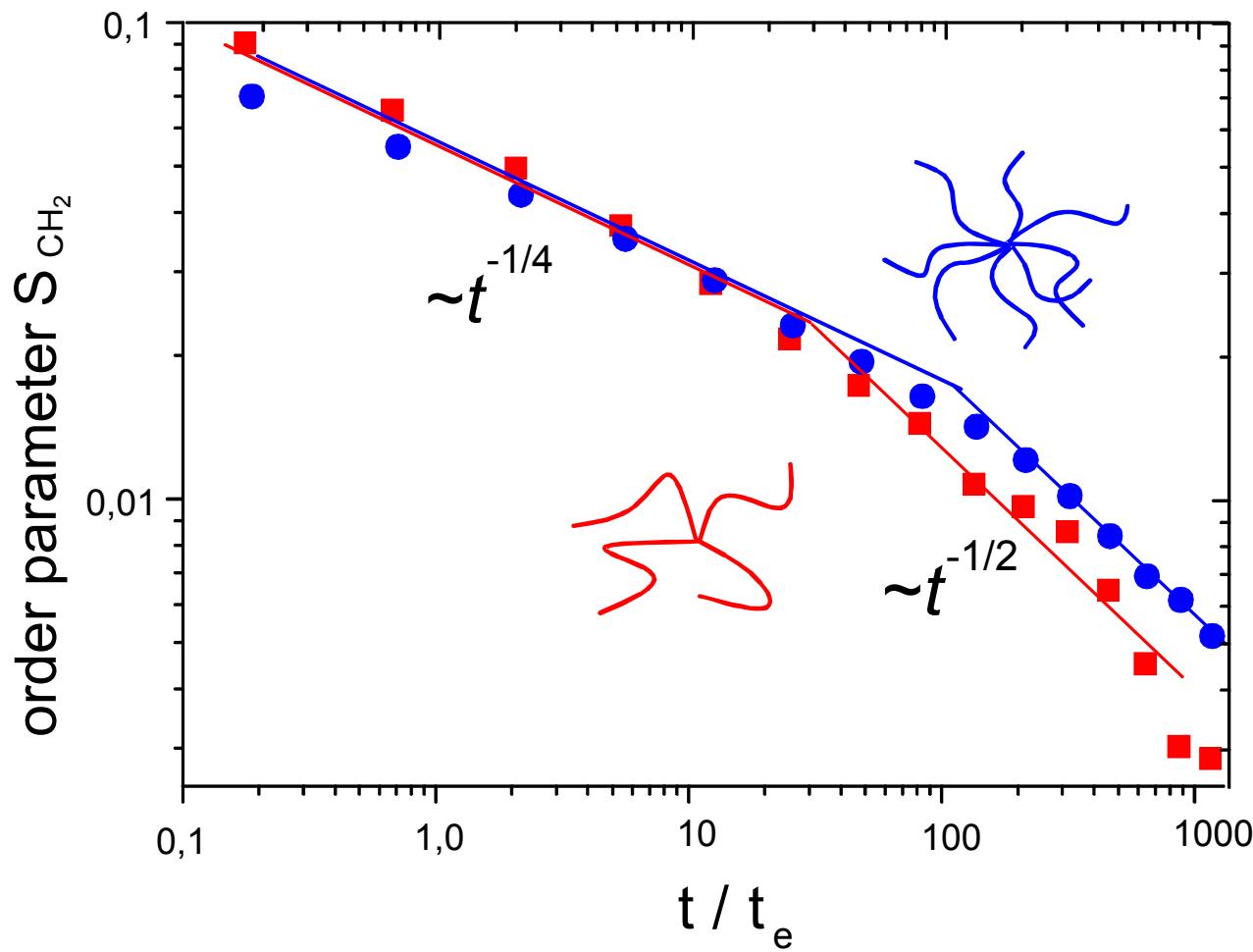
PB(d_6)-PB-PB(d_6)



PB

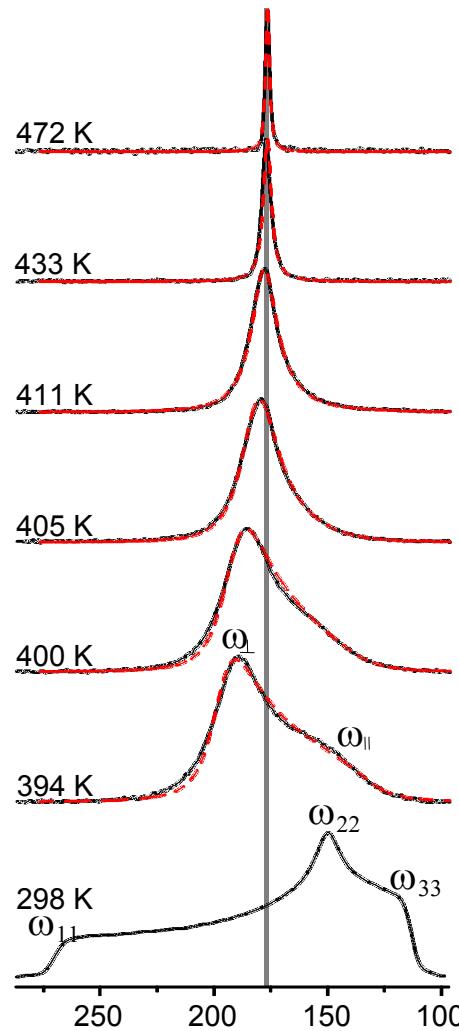
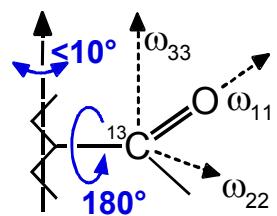
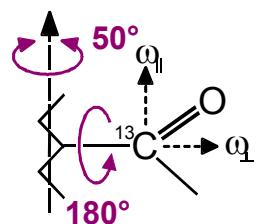
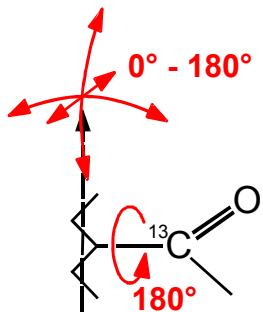
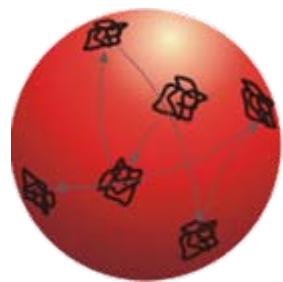


Local Order Parameters in Star Shaped Polymers



cooperation with Prof. Hadjichristidis / Athen.

a-PEMA: Isotropisation of Chain Dynamics



1D ^{13}C NMR: — experiment
- - - simulation

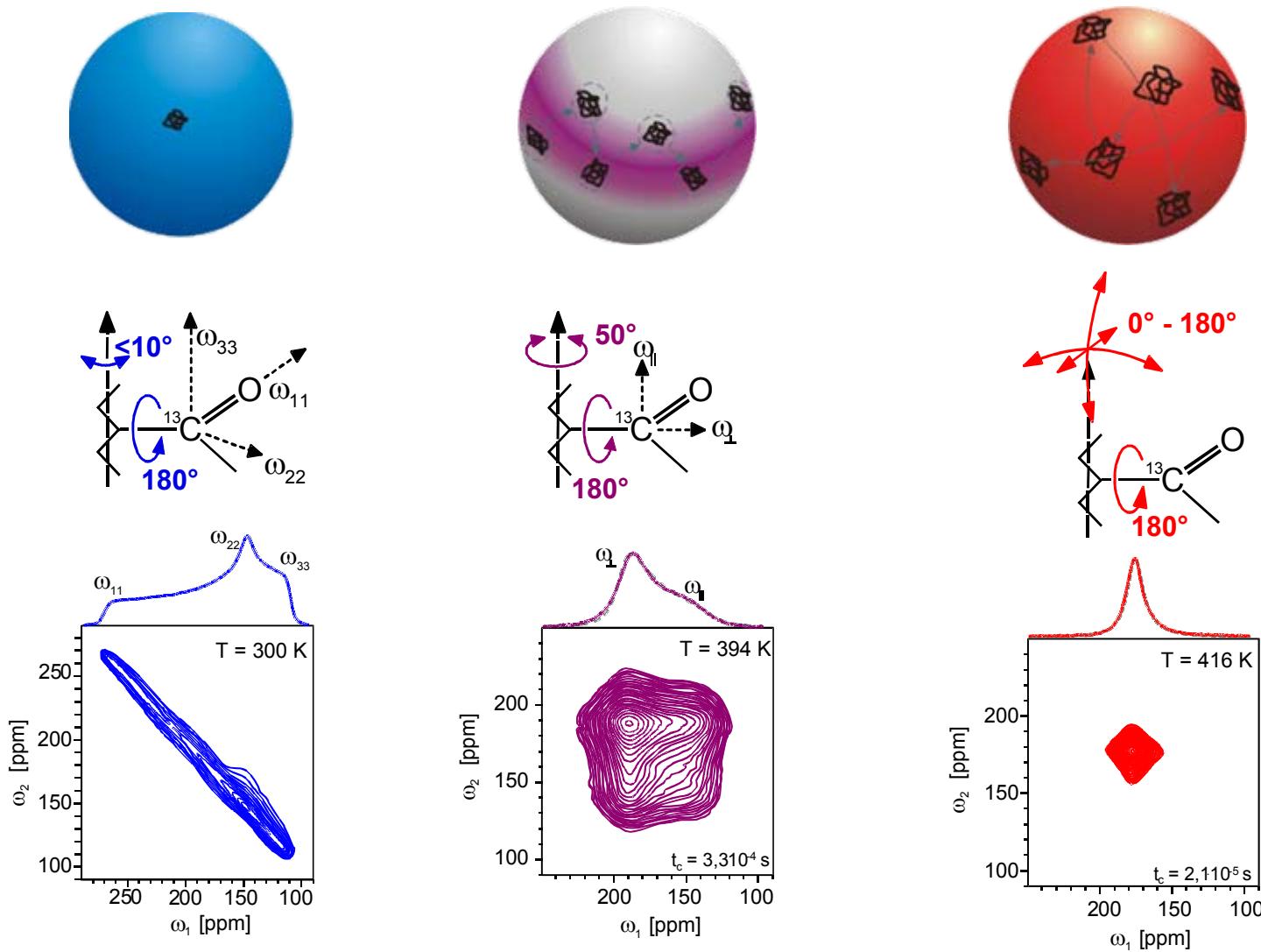
Melt

Melt

Glass
 T_g



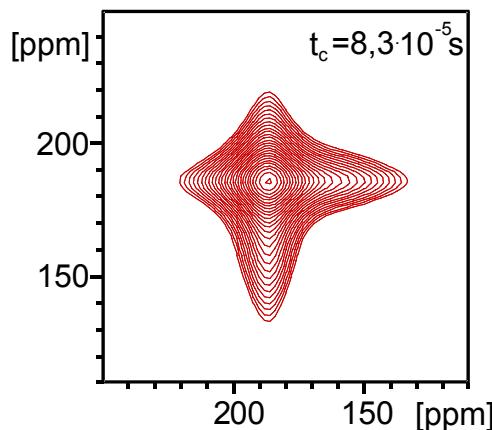
a-PEMA: Isotropisation of Chain Dynamics



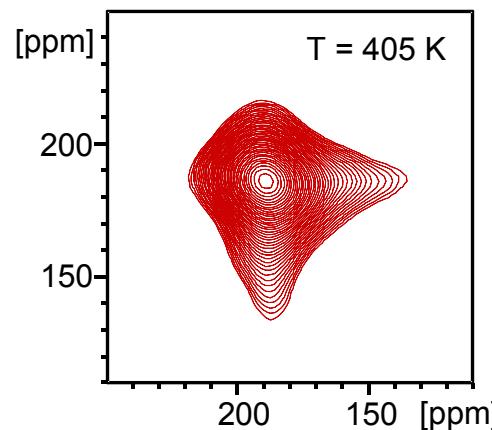
Dynamic Models: Random Jump vs. Rotational Diffusion



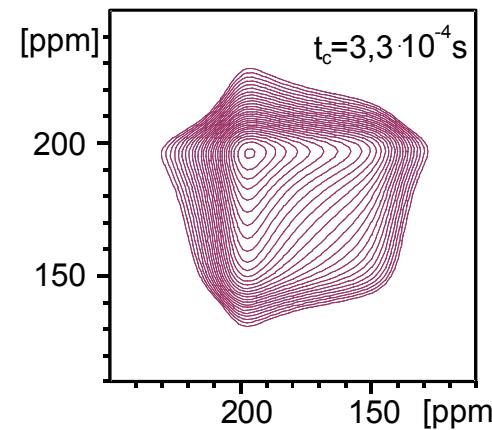
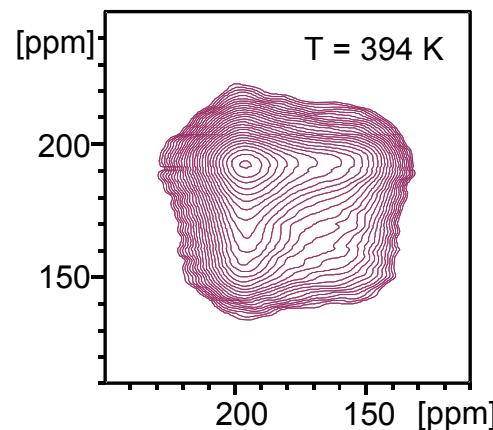
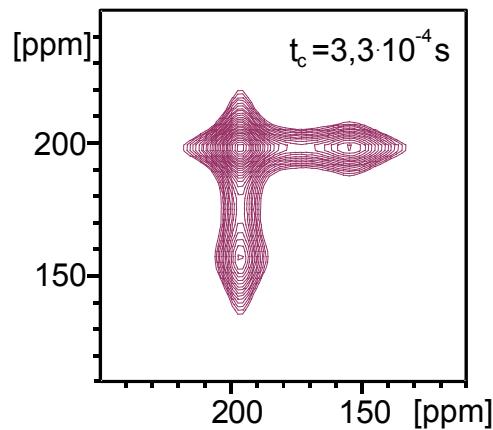
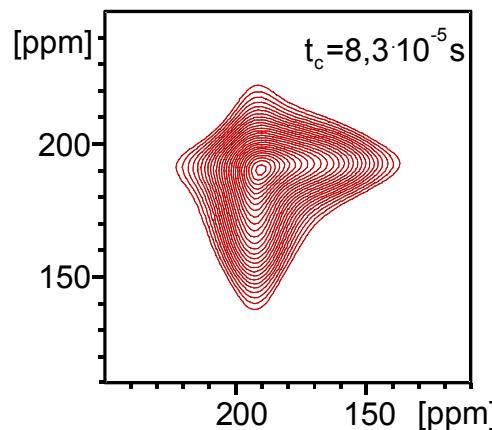
rotational diffusion



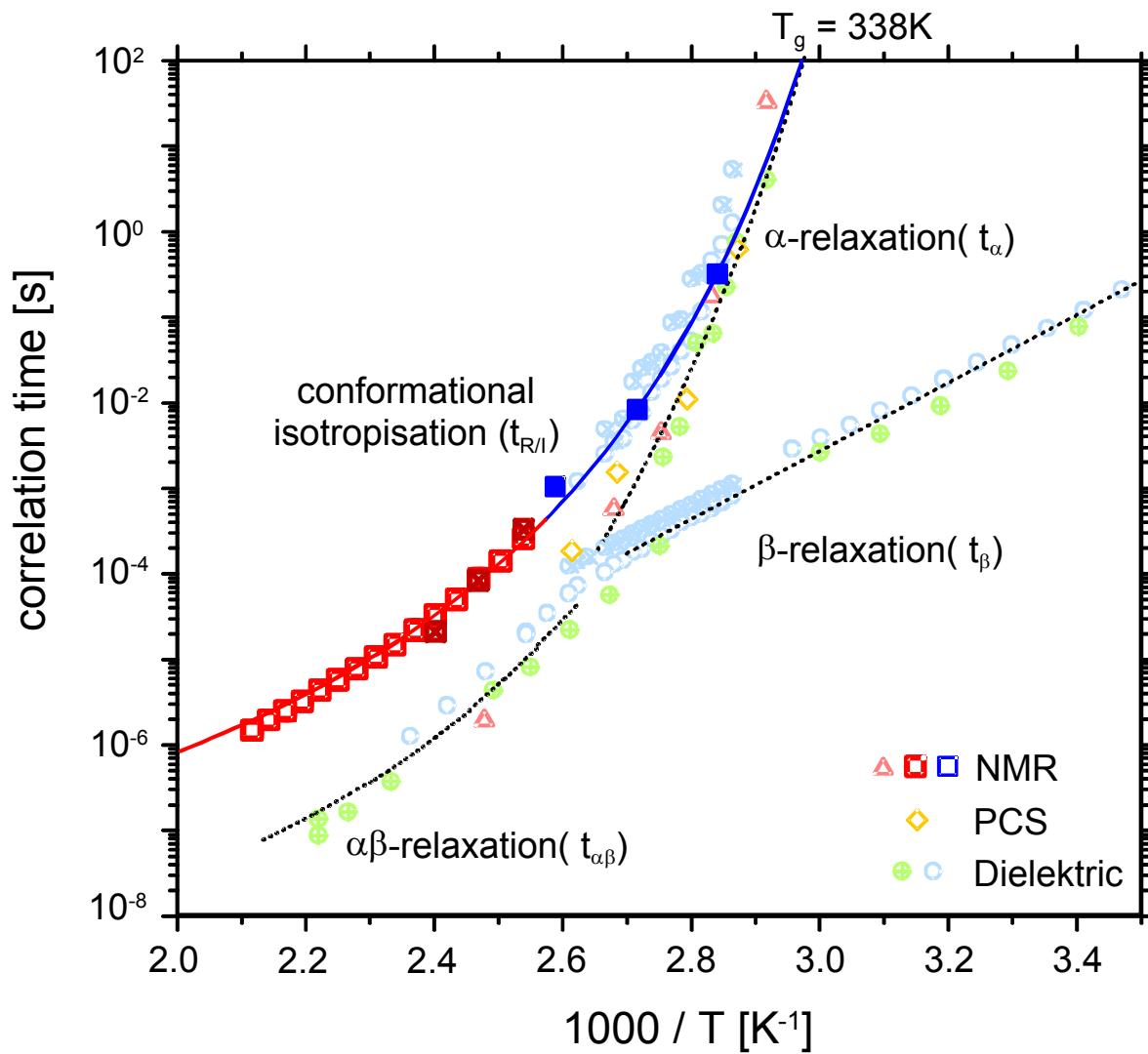
experimental results



random jump



Time Scales of Molecular Dynamics PEMA Melts



Arrhenius-diagram of dynamic processes in PEMA

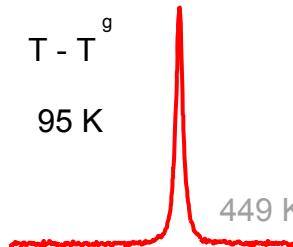
Length Scale of Isotropisation Process



from radicalic polymerisation

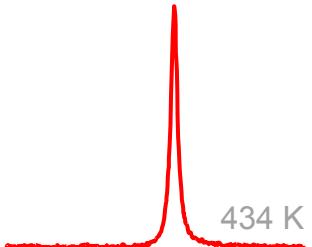
$$M_w = 600 \text{ kg mol}^{-1}$$

$$M_w/M_n = 1,53$$



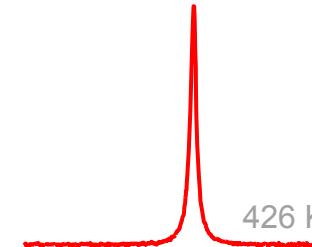
$$M_w = 15,9 \text{ kg mol}^{-1}$$

$$M_w/M_n = 1,61$$



$$M_w = 7,6 \text{ kg mol}^{-1}$$

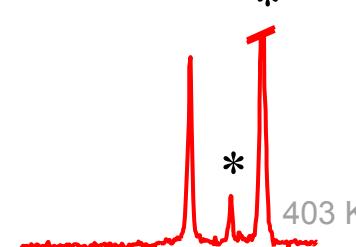
$$M_w/M_n = 1,48$$



from anionic polymerisation

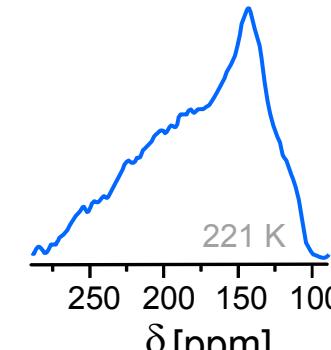
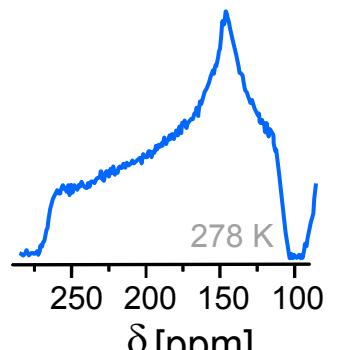
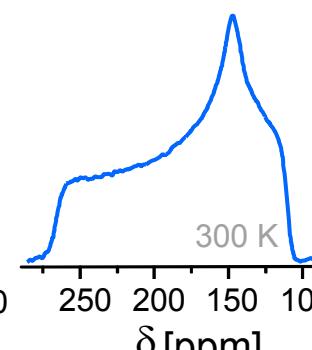
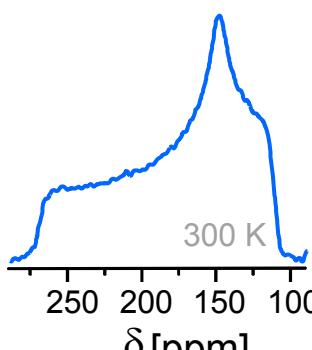
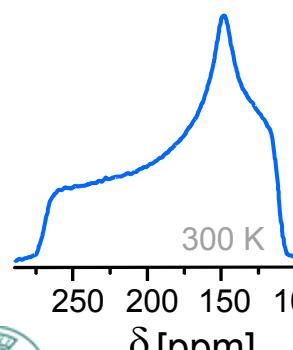
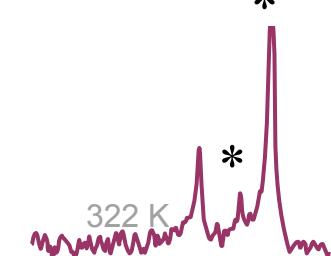
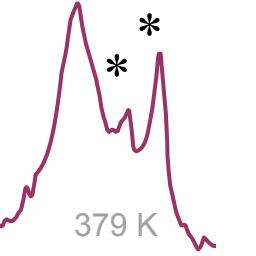
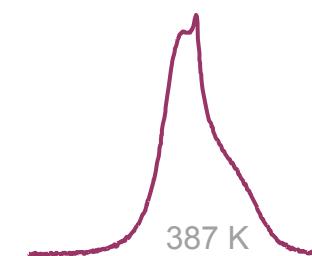
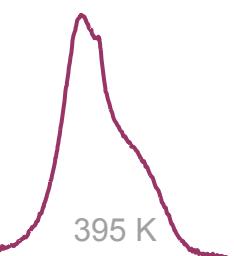
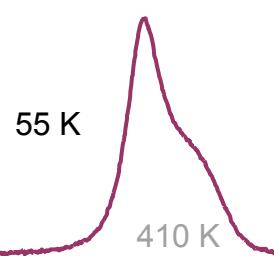
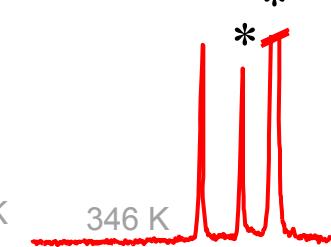
$$M_w = 1,4 \text{ kg mol}^{-1}$$

$$M_w/M_n = 1,07^*$$

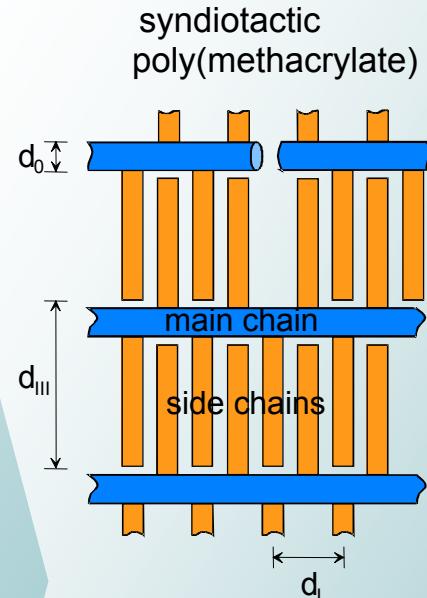
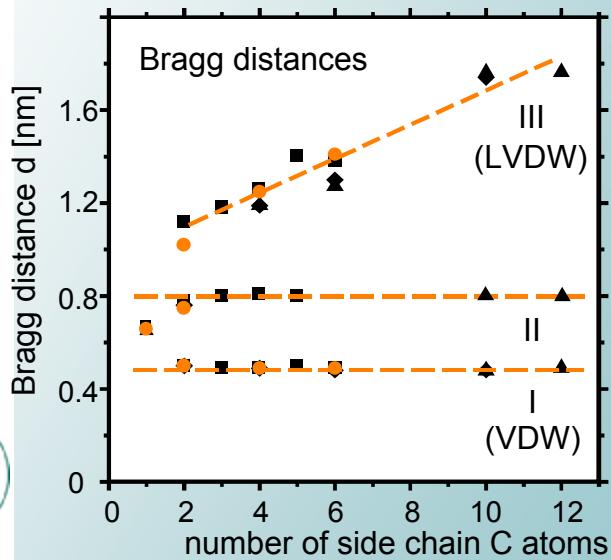
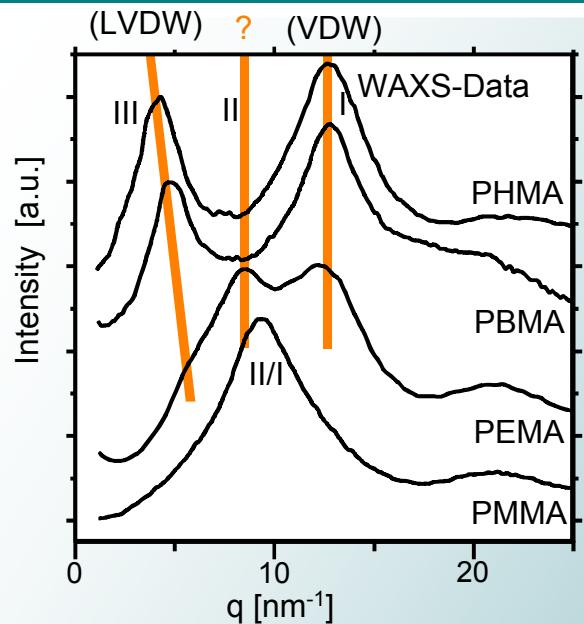


$$M_w = 0,46 \text{ kg mol}^{-1}$$

$$M_w/M_n = 1,14^*$$

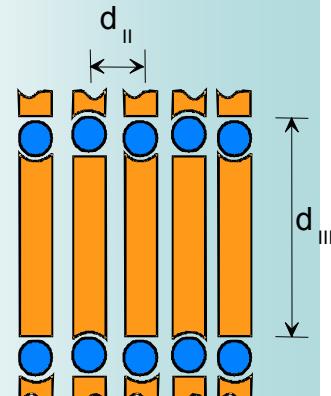
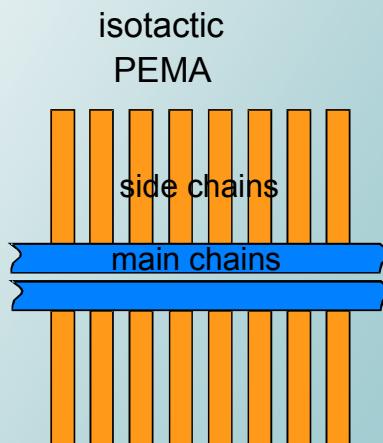


Organisation in Poly(Methacrylates): WAXS



extrapolated
lokal structur:

"Nano Layers"





How Unstructured are Amorphous Polymer Melts? Solid-State NMR Studies of Local Dynamic Order in Amorphous Polymer Melts

Introduction • Interaction in solid state NMR

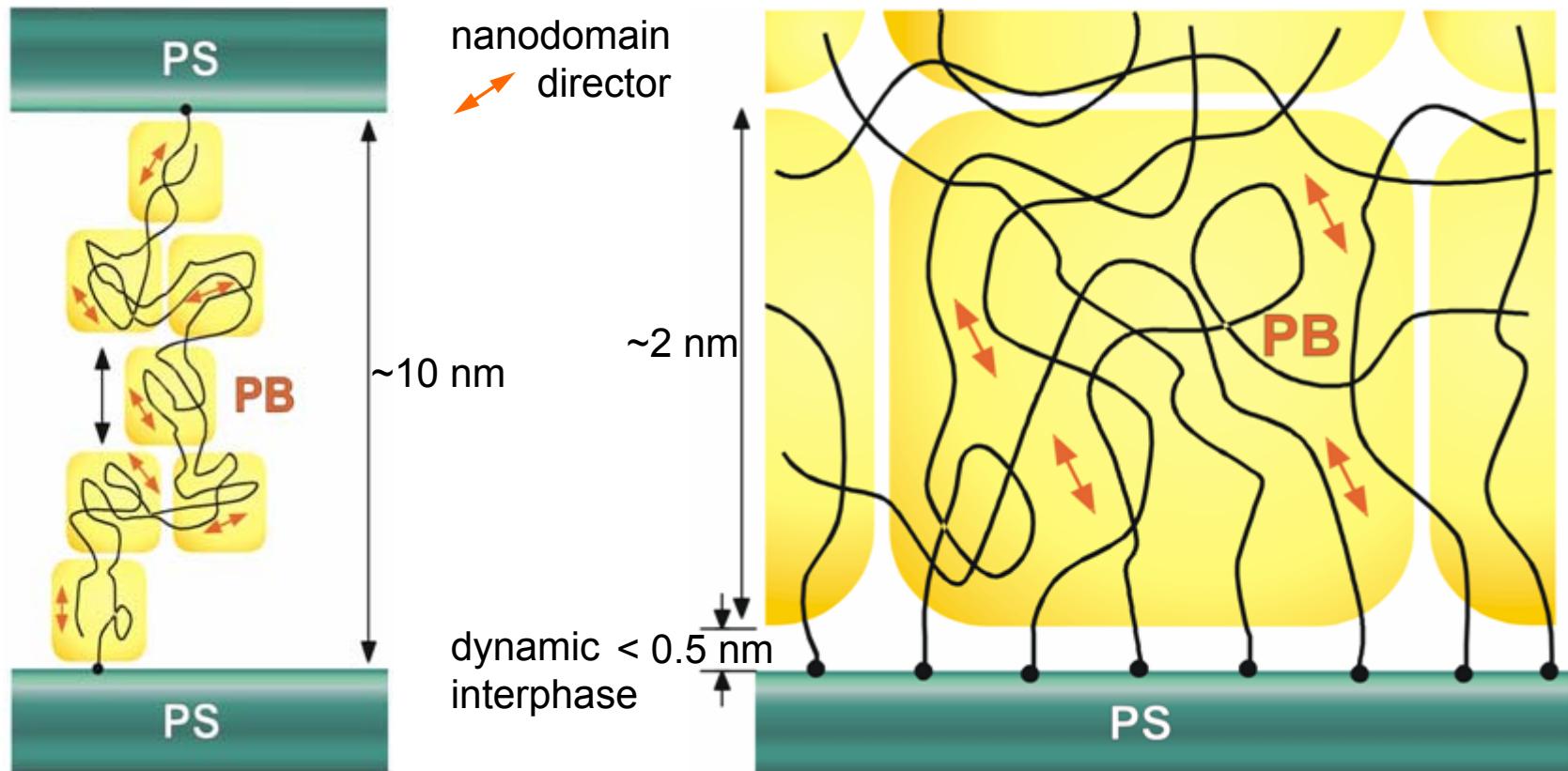
Solid State NMR • MAS, recoupling, double-quantum NMR

Polymer Dynamics • Reptation-model, polybutadiene, PEMA

Conclusions • How unstructured are amorphous polymers ?



Längenskalen lokaler Ordnung in Polymerschmelzen



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