



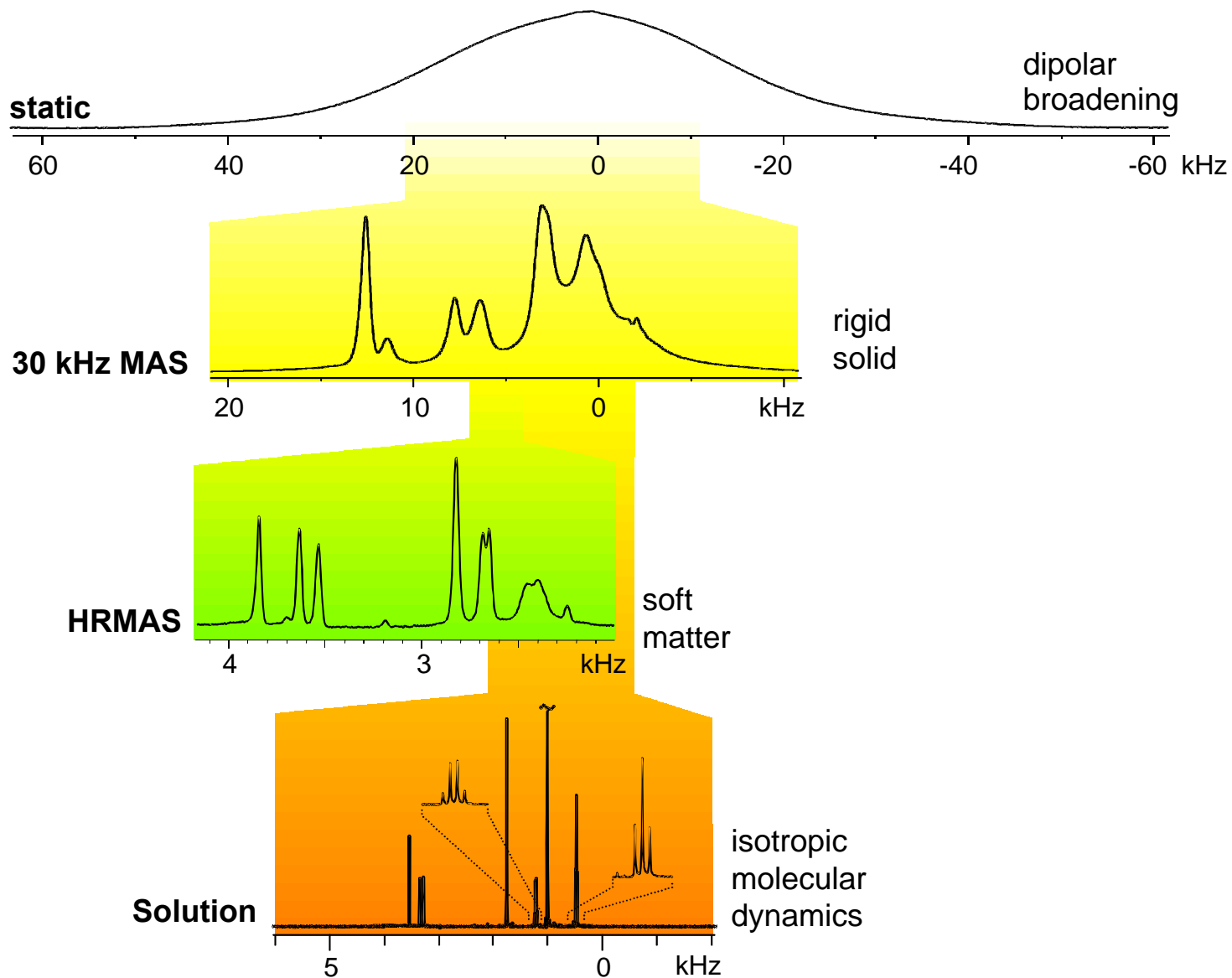
NMR Studies of Polyethylene: Towards the Organization of Semi Crystalline Polymers

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Leibniz Institut für Polymerforschung, Dresden, January 15th 2008

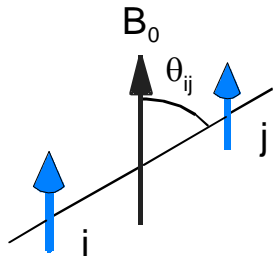
Line Width in ^1H NMR Spectra



Resolution Enhancement in NMR



dipole-dipole coupling:



$$\hat{H} = \hat{R}_{2,0} \cdot \hat{T}_{2,0}$$

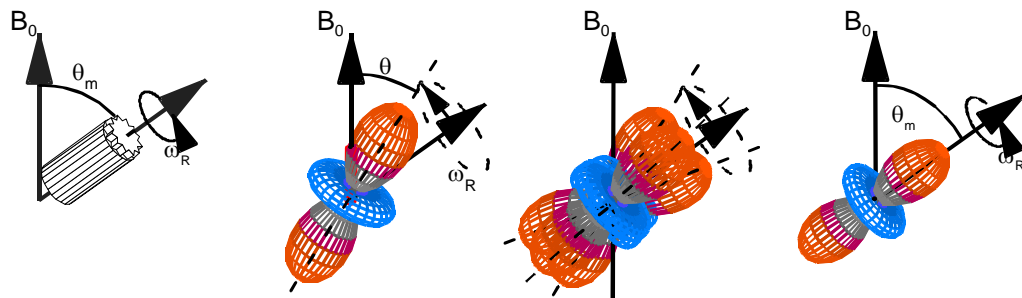
space

spin

$$\hat{H} \propto \frac{1}{r_{ij}^3} \frac{1}{2} (3 \cos^2 \theta_{ij} - 1) \gamma_i \gamma_j (3 \hat{I}_{Z,i} \hat{I}_{Z,j} - \hat{I}_i \cdot \hat{I}_j)$$

magic angle spinning:

$$\hat{R}_{2,0} \rightarrow 0$$

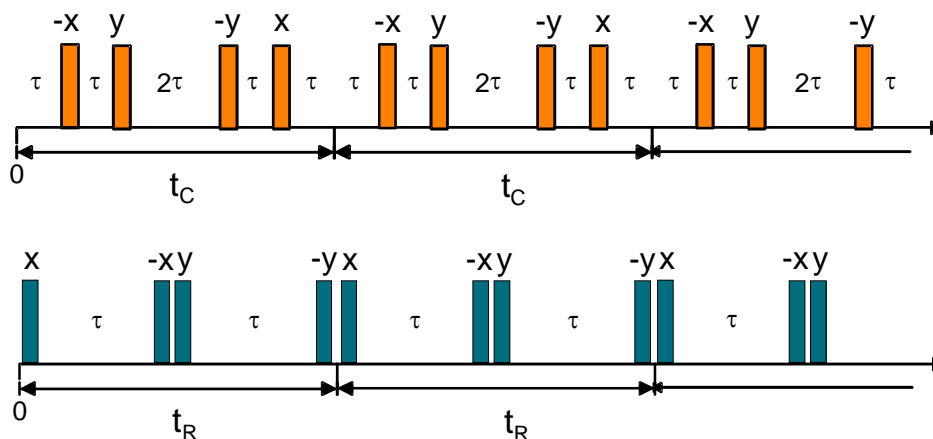


RF irradiation:

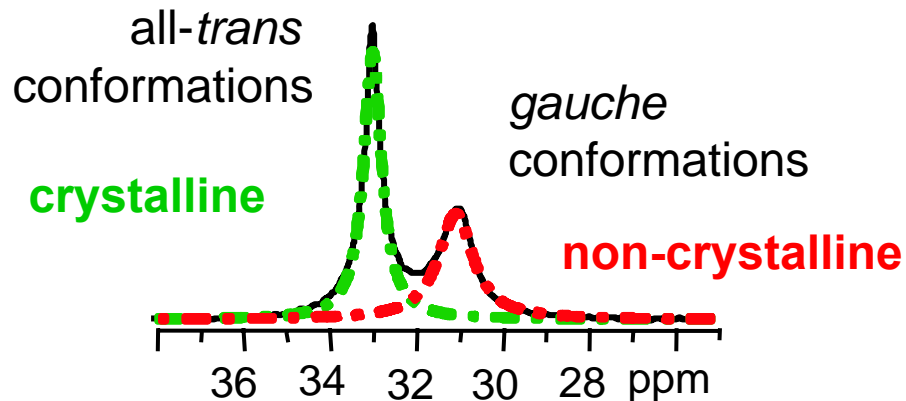
$$\hat{T}_{2,0} \rightarrow 0 \quad \text{(CRAMPS)}$$



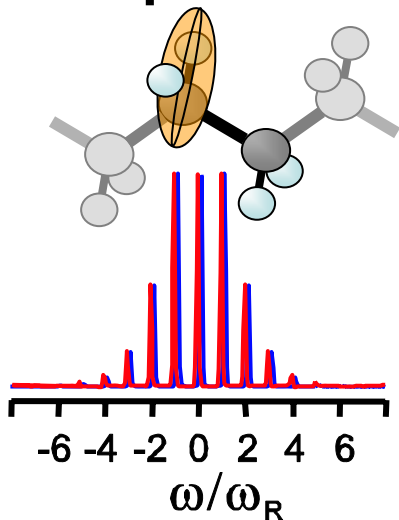
$$H_{D,eff.} \text{ (Recoupling)}$$



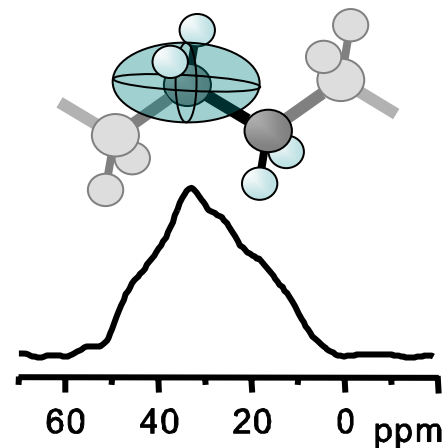
Anisotropic NMR Interactions



^1H - ^{13}C dipolar couplings



^{13}C chemical shift anisotropy



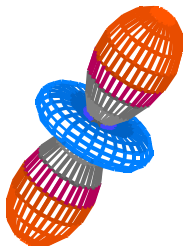
Dipolar Sideband Pattern



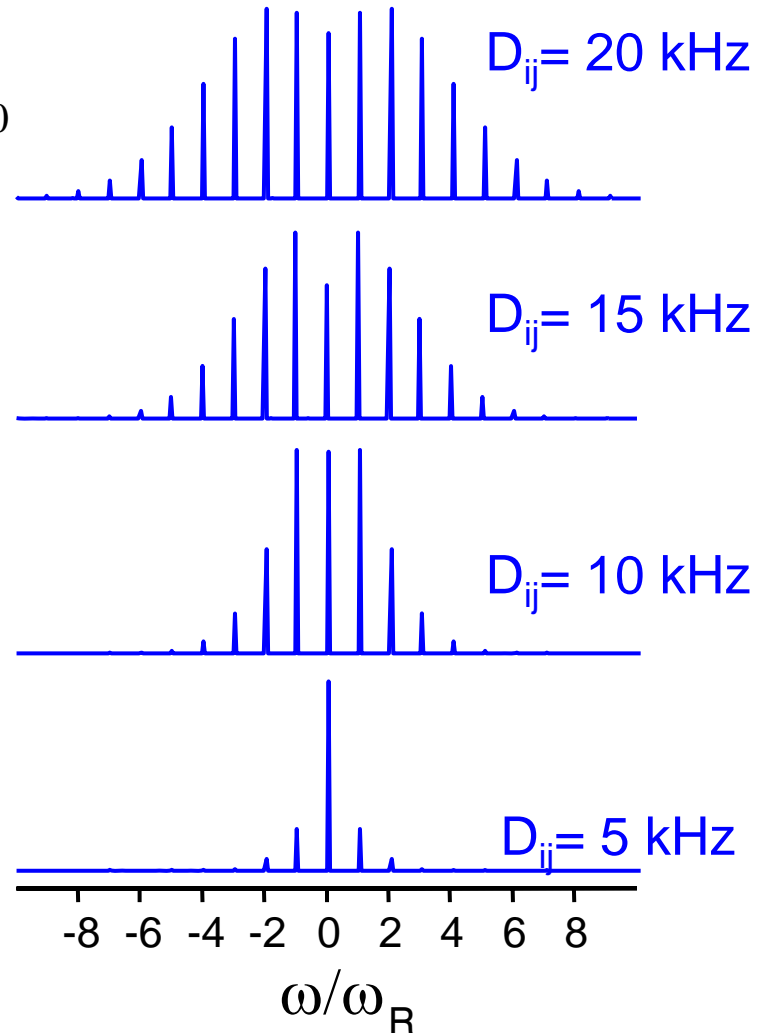
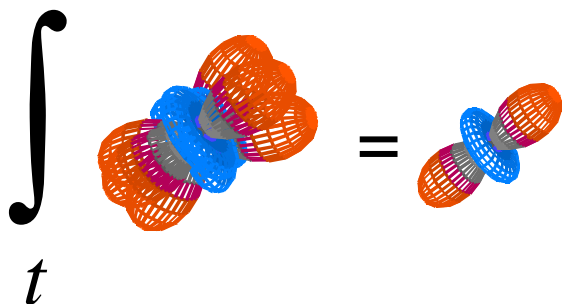
dipolar couplings:

ReREDOR sideband pattern

$$H_D = \sum_{i \neq j} \frac{\mu_0 \hbar}{4\pi} \frac{\gamma_i \gamma_j}{r^3} \frac{1}{2} (3 \cos^2 \theta_{ij} - 1) \mathbf{T}_{2,0}^{ij}$$



motional averaged



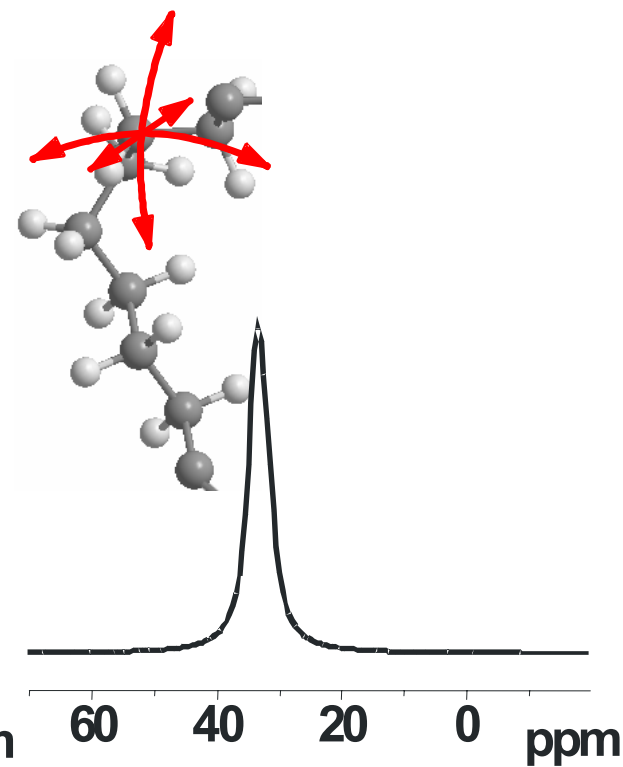
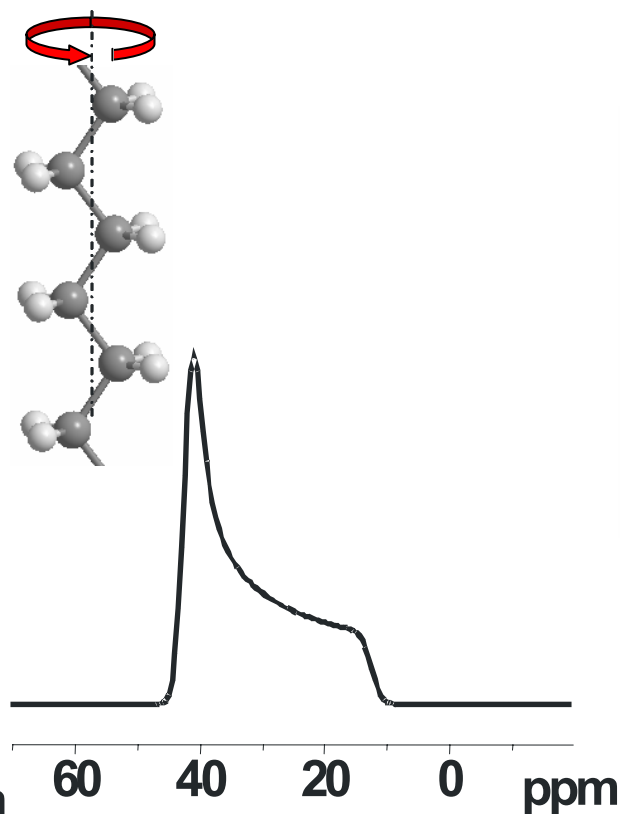
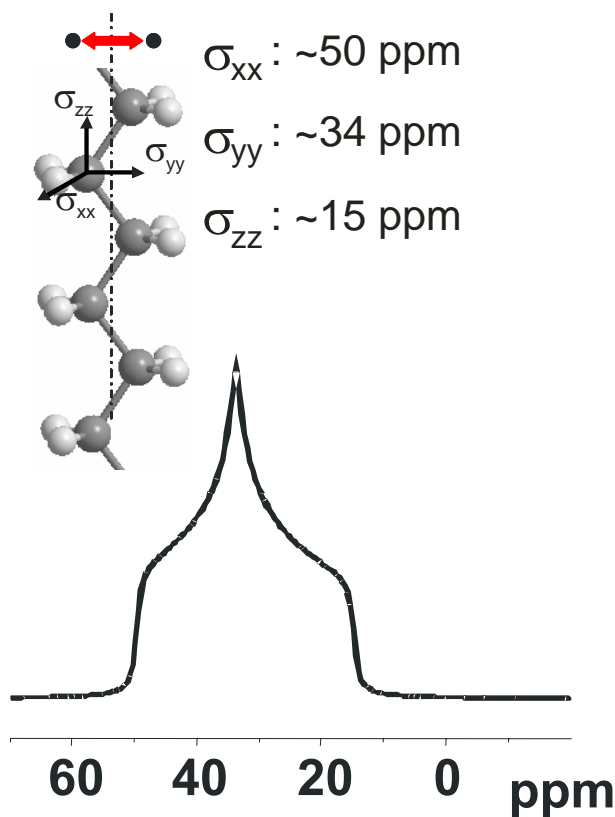
Chemical Shift Anisotropy Pattern



180° hopping

axial rotation

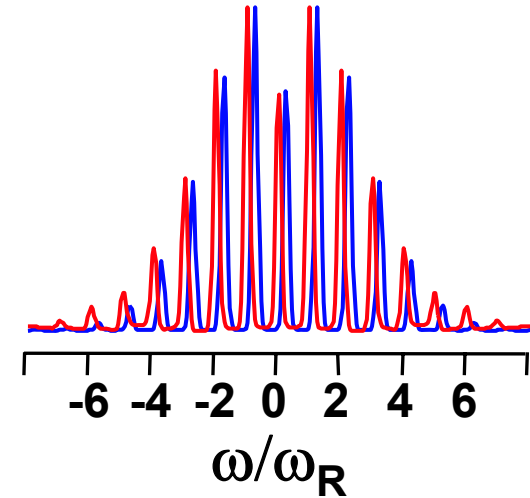
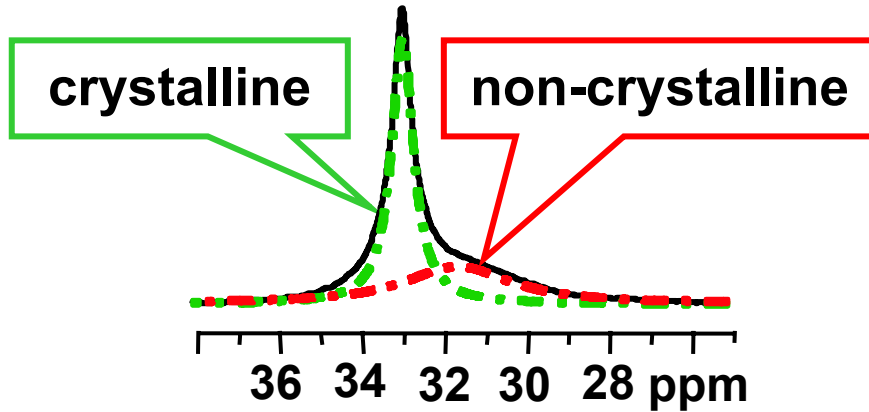
isotropic motion



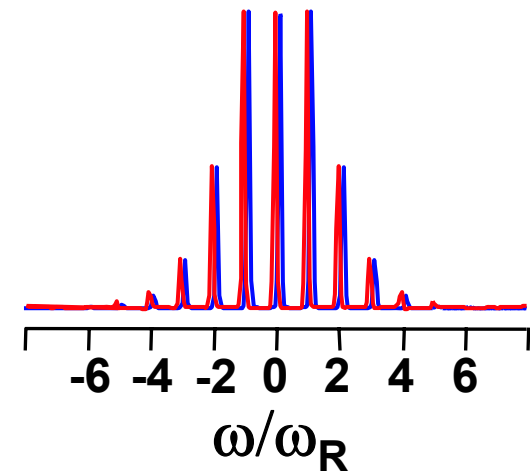
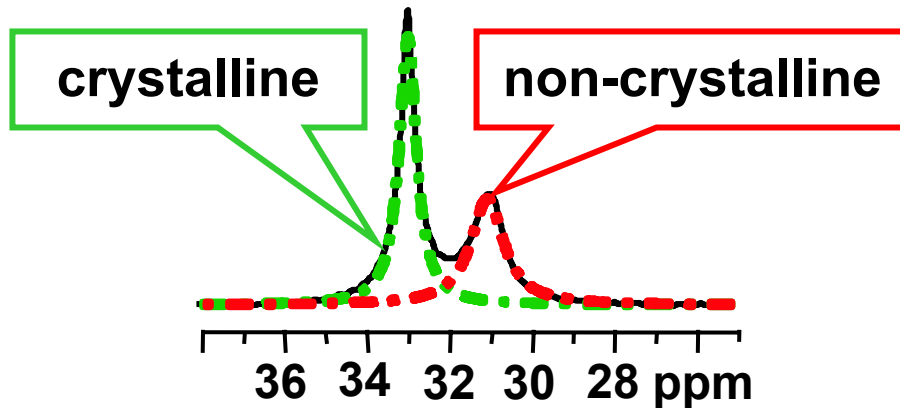
Morphology and Chain Dynamics in PE



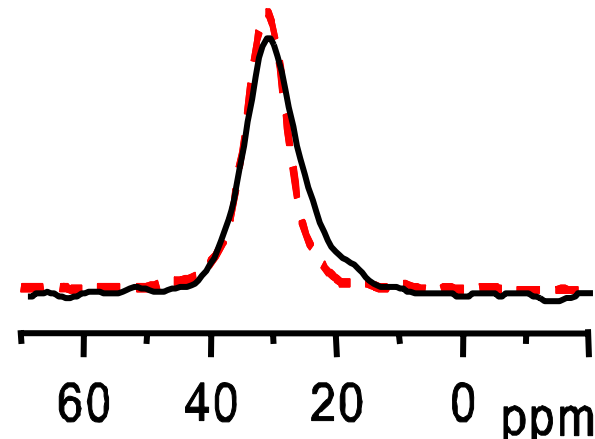
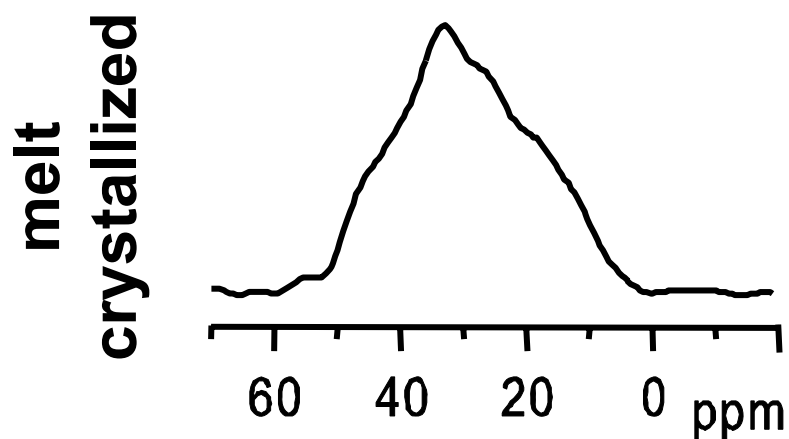
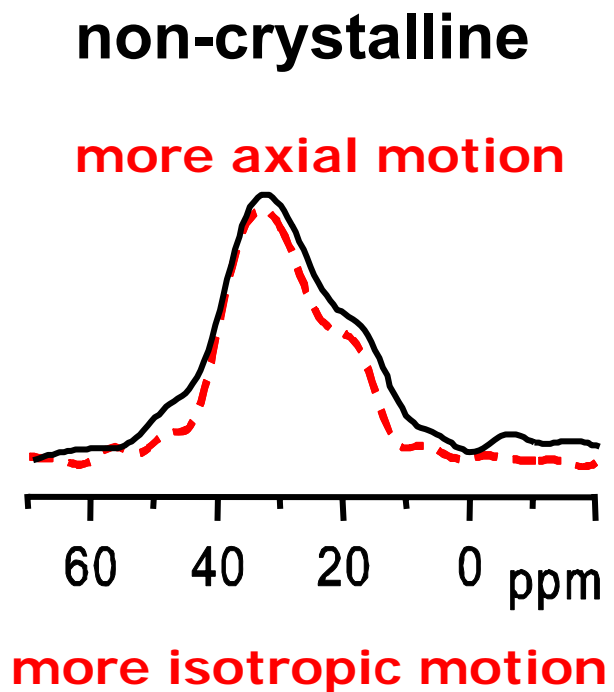
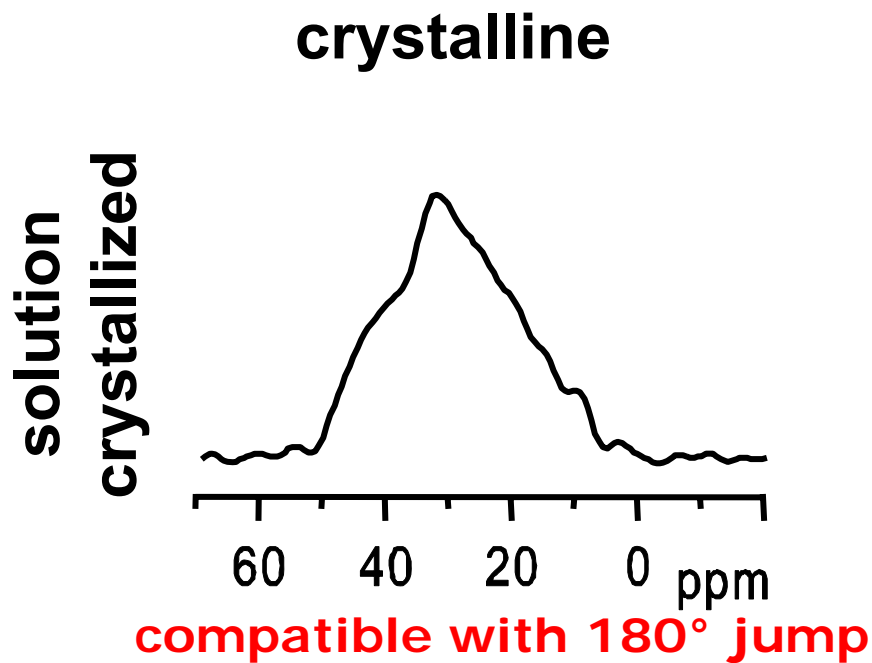
solution crystallized ultra high molecular weight PE



melt crystallized ultra high molecular weight PE



^{13}C Chemical Shift Anisotropy

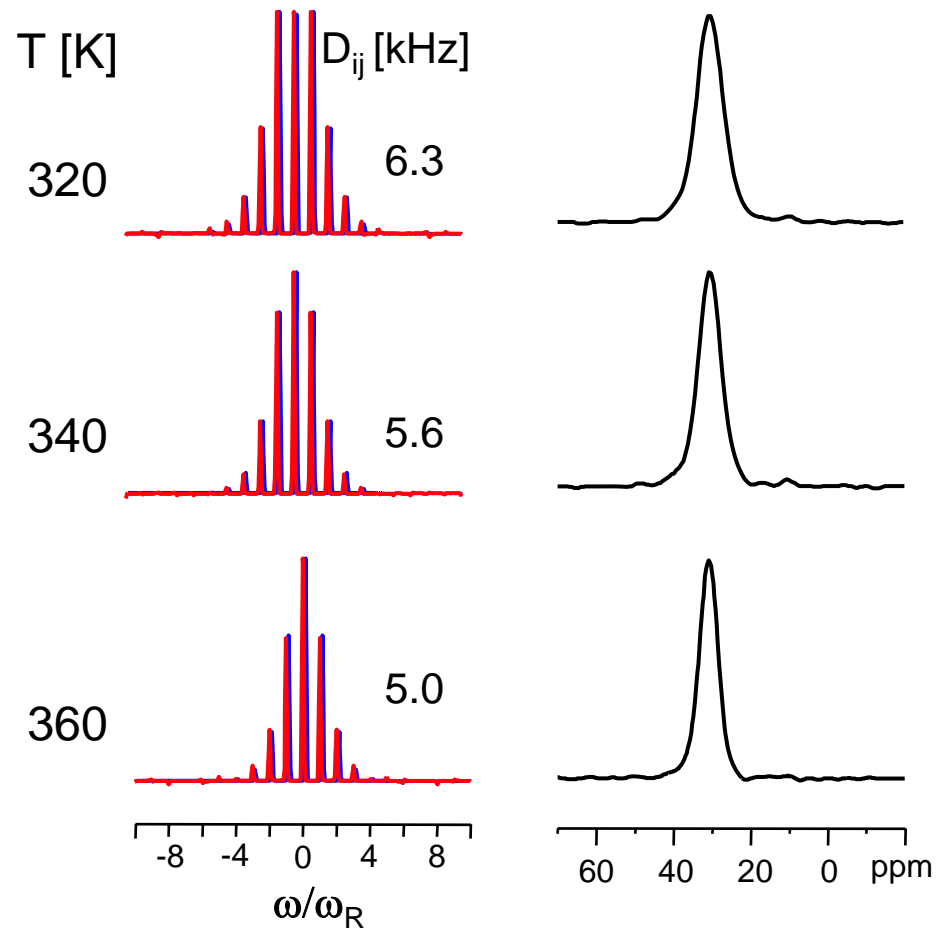
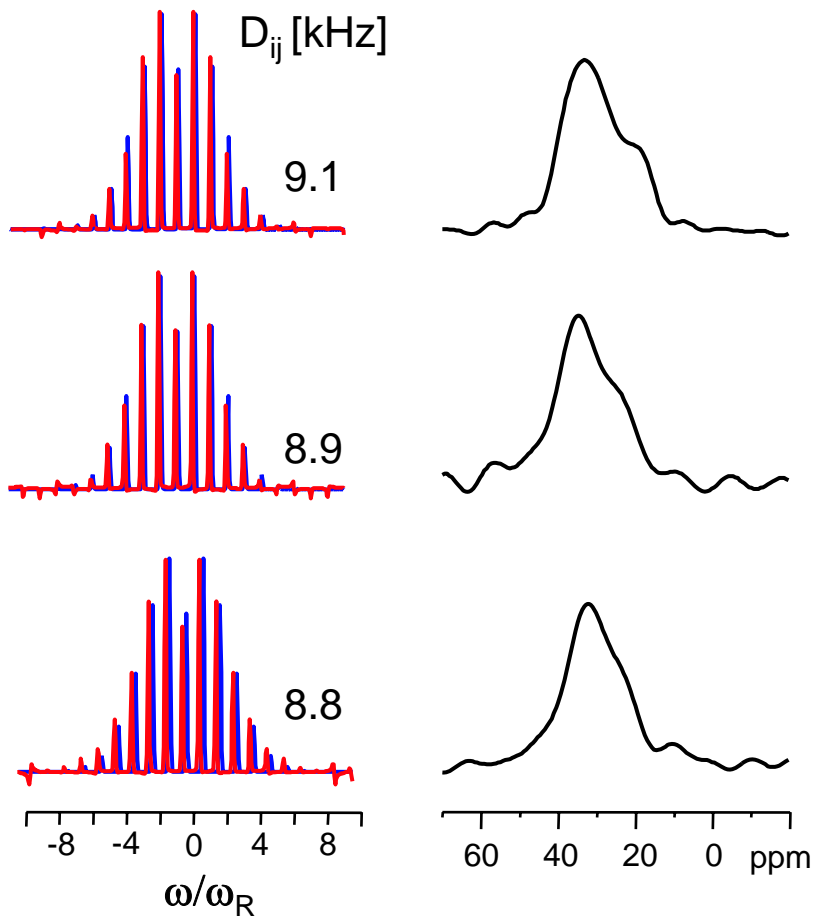


Temperature Dependence



solution crystallized UHMW-PE

melt crystallized UHMW-PE





Observation and Analysis of Chain Translation Motion with ^{13}C NMR Experiments

Observation of Chain Translation



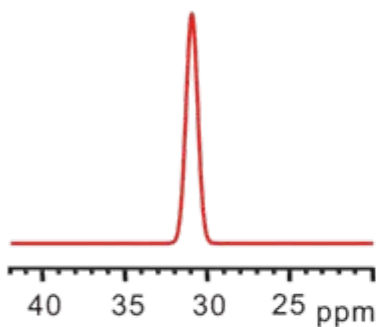
saturate ^{13}C

^{13}C relaxation / chain diffusion

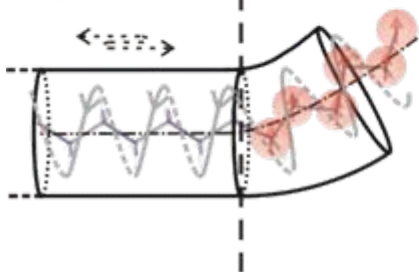
acquisition

$t \approx 0$

non-crystalline

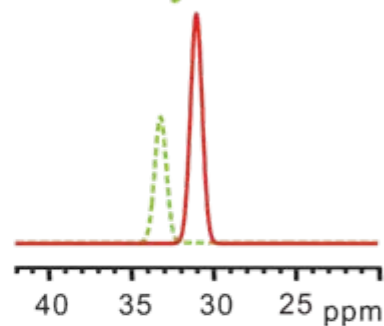


crystalline | non-crystalline

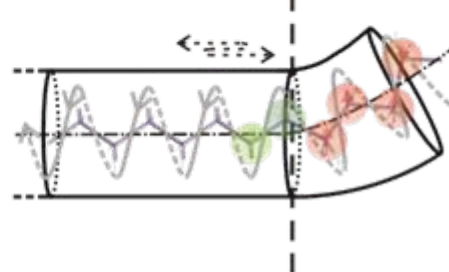


$t < 0$

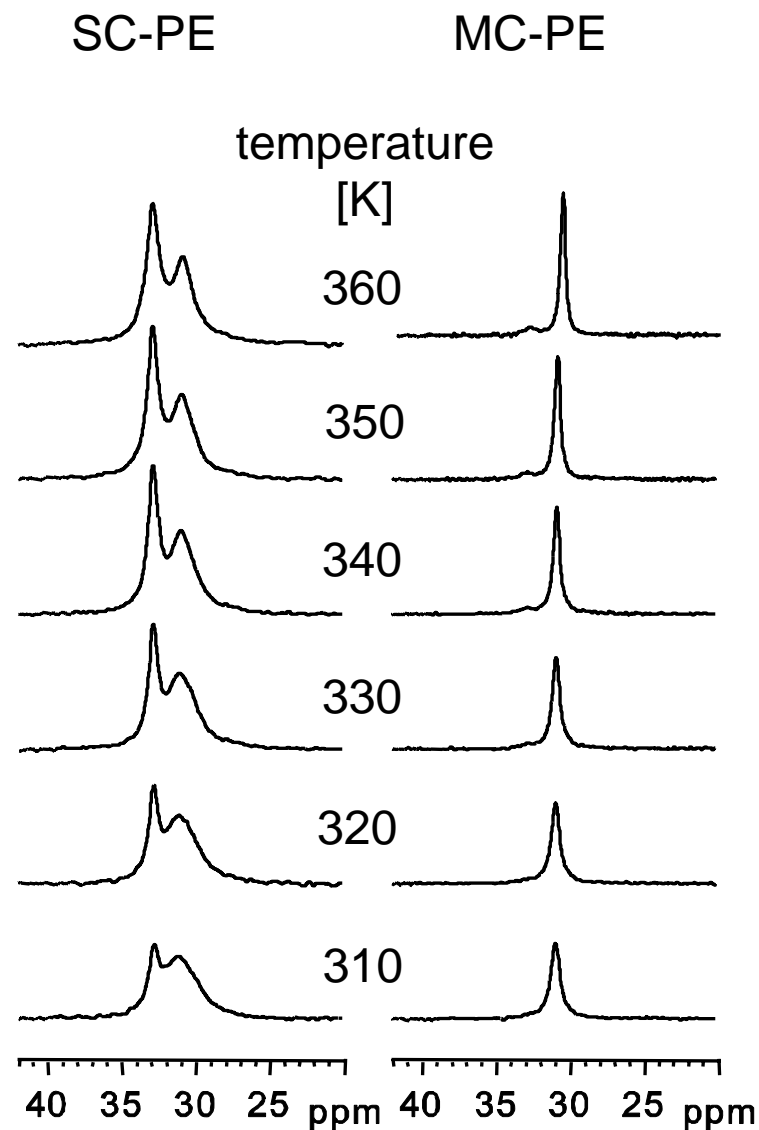
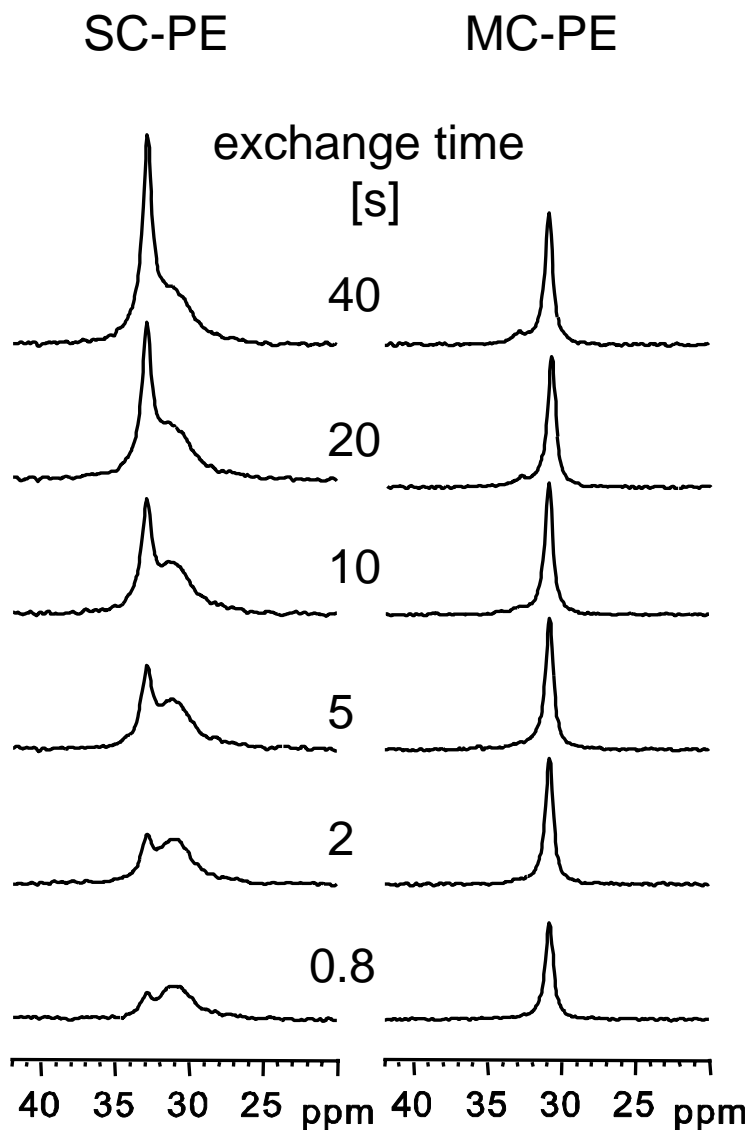
non-crystalline
and crystalline



crystalline | non-crystalline



Time and Temperature Dependence



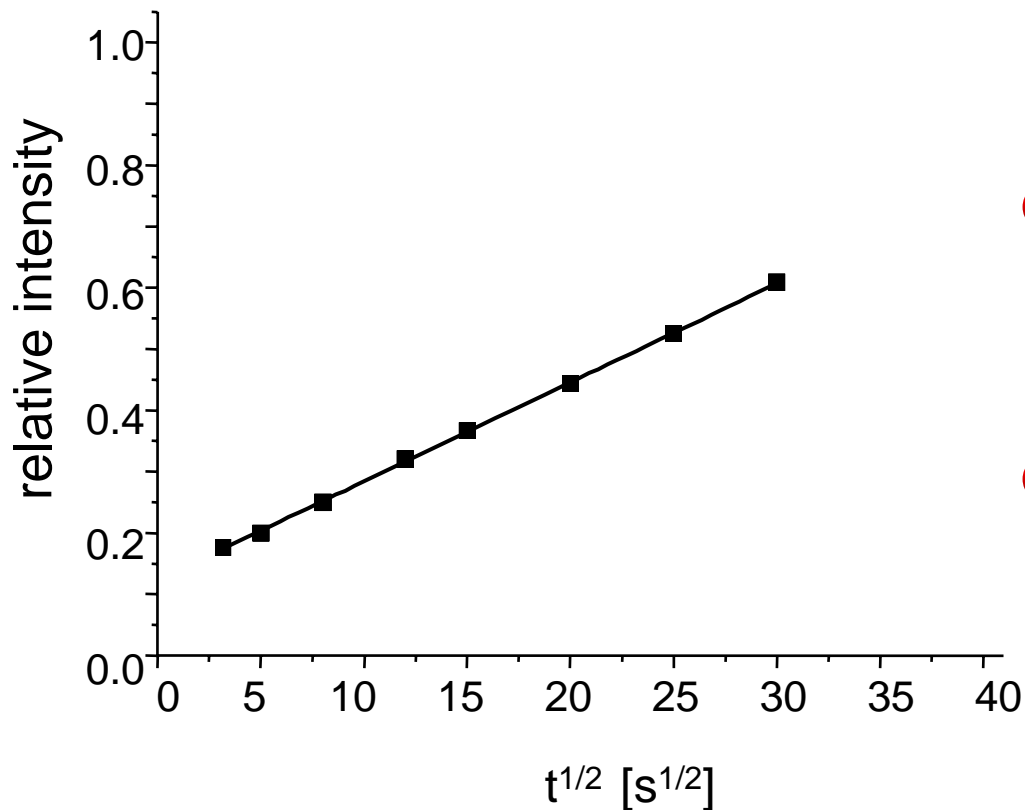
Quantification of Chain Translation



saturate ^{13}C

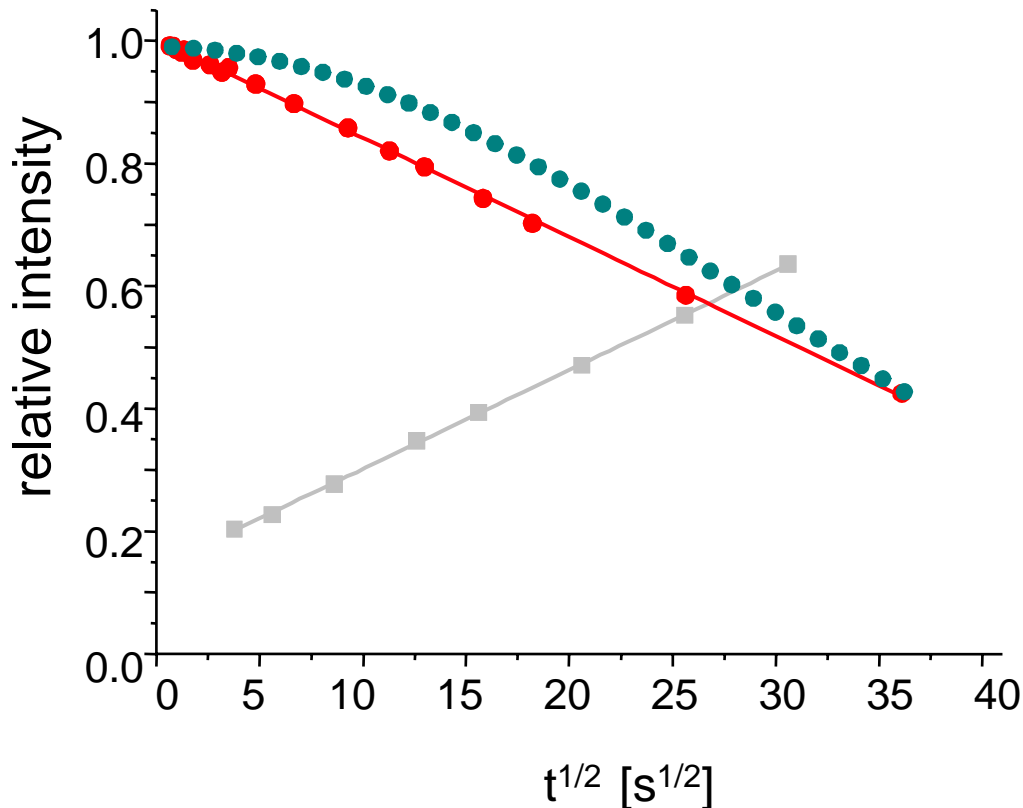
^{13}C relaxation / chain diffusion

acquisition



- Intensities at short times are affected by T_1 relaxation
- Normalization needs the Intensity at infinite diffusion time.

Quantification of Chain Translation

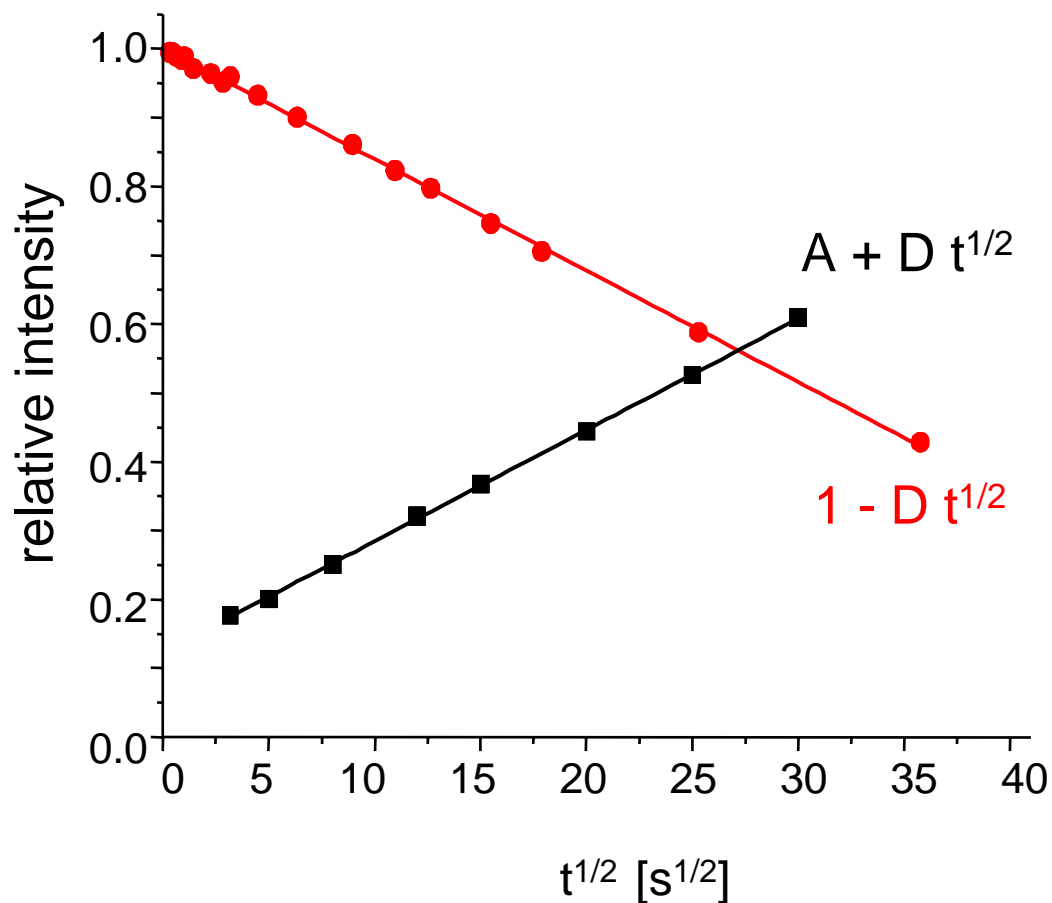


~~● Intensities at short times are affected by T_1 relaxation~~

~~● Normalization needs the Intensity at infinite diffusion time.~~

● Non-exponential signal decay does not originate from T_1 relaxation

Determination of NMR Crystallinity

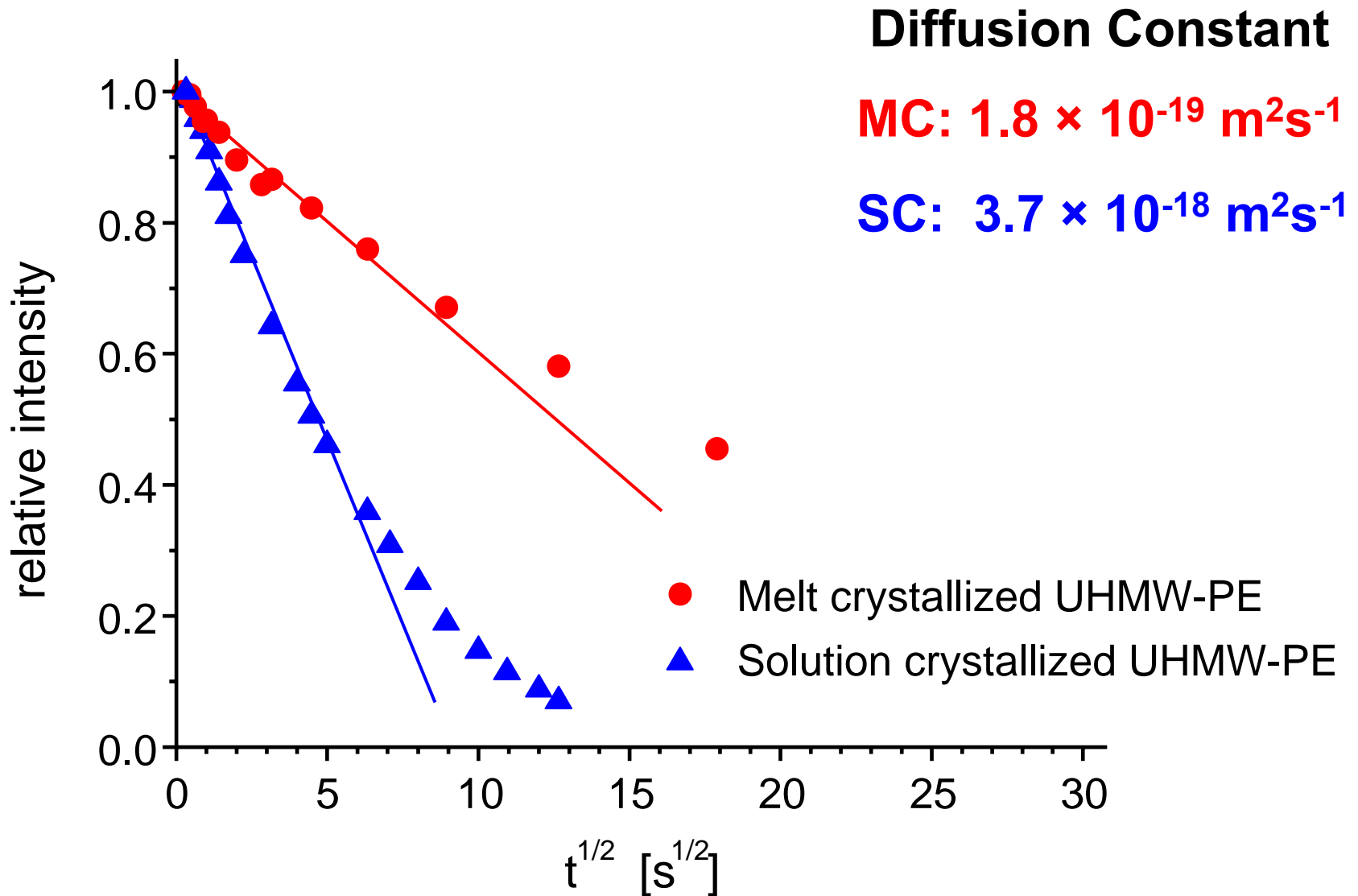


crystallinity C:

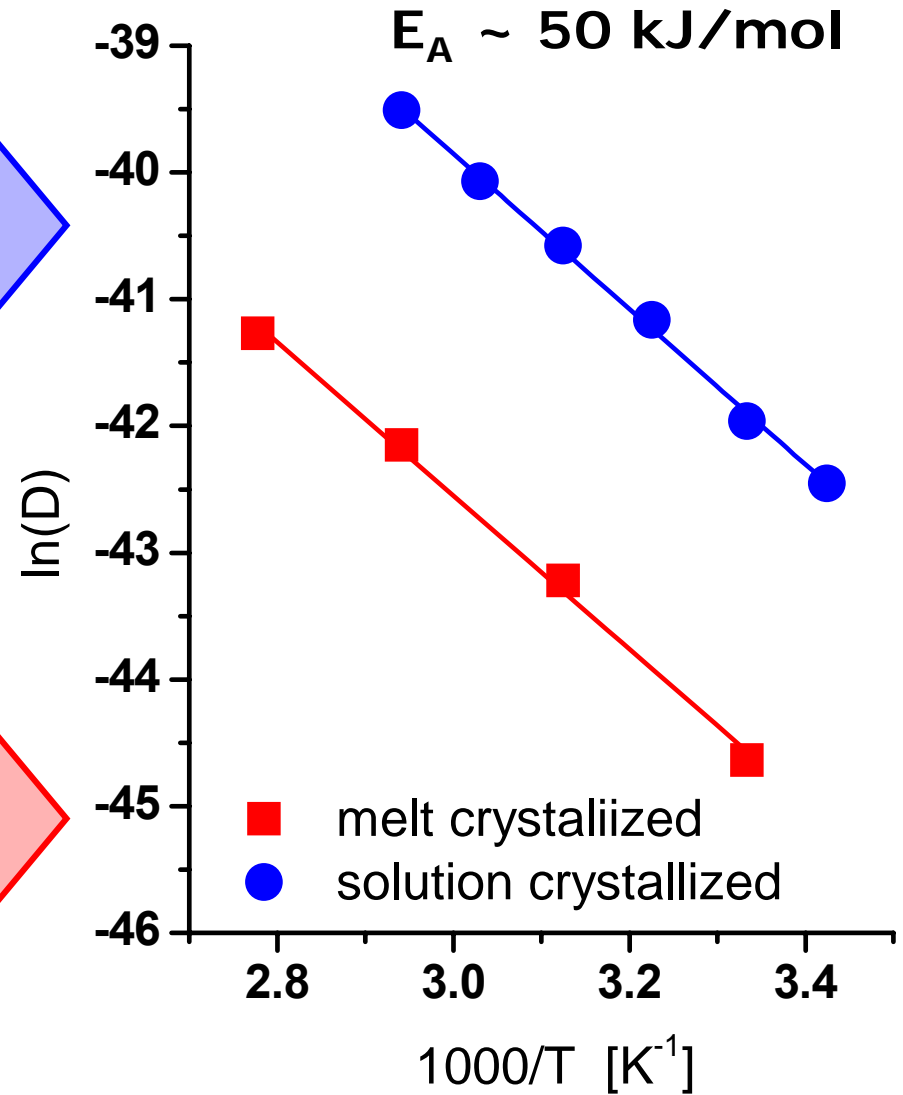
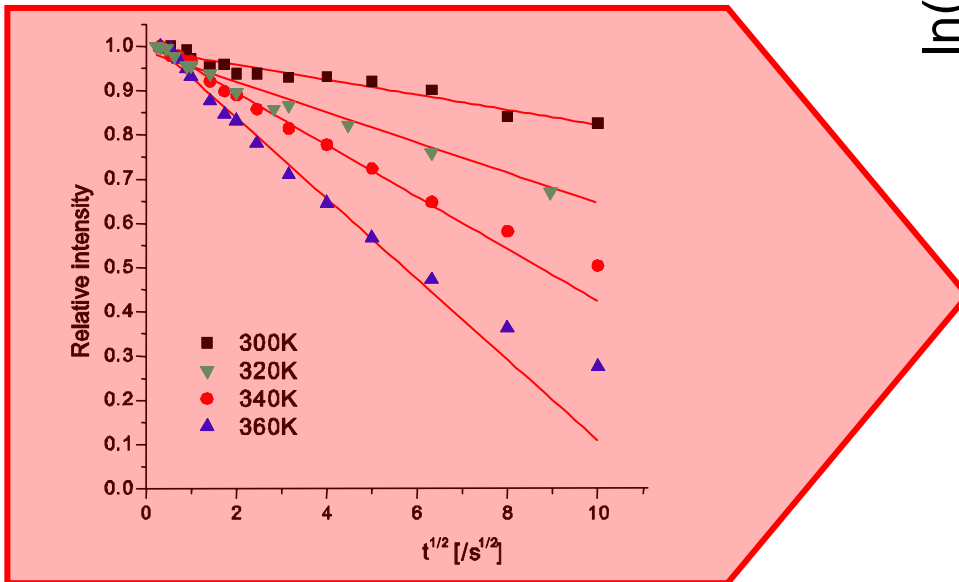
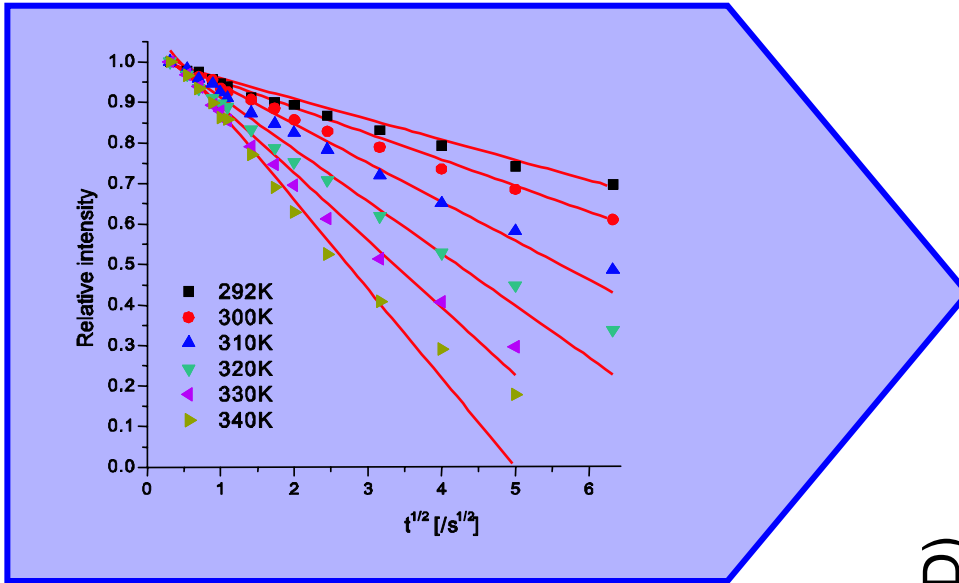
$$C = 1 / (A+1)$$

	x-ray	NMR
MC	39%	46%
SC	74%	75%
fiber	95%	90%

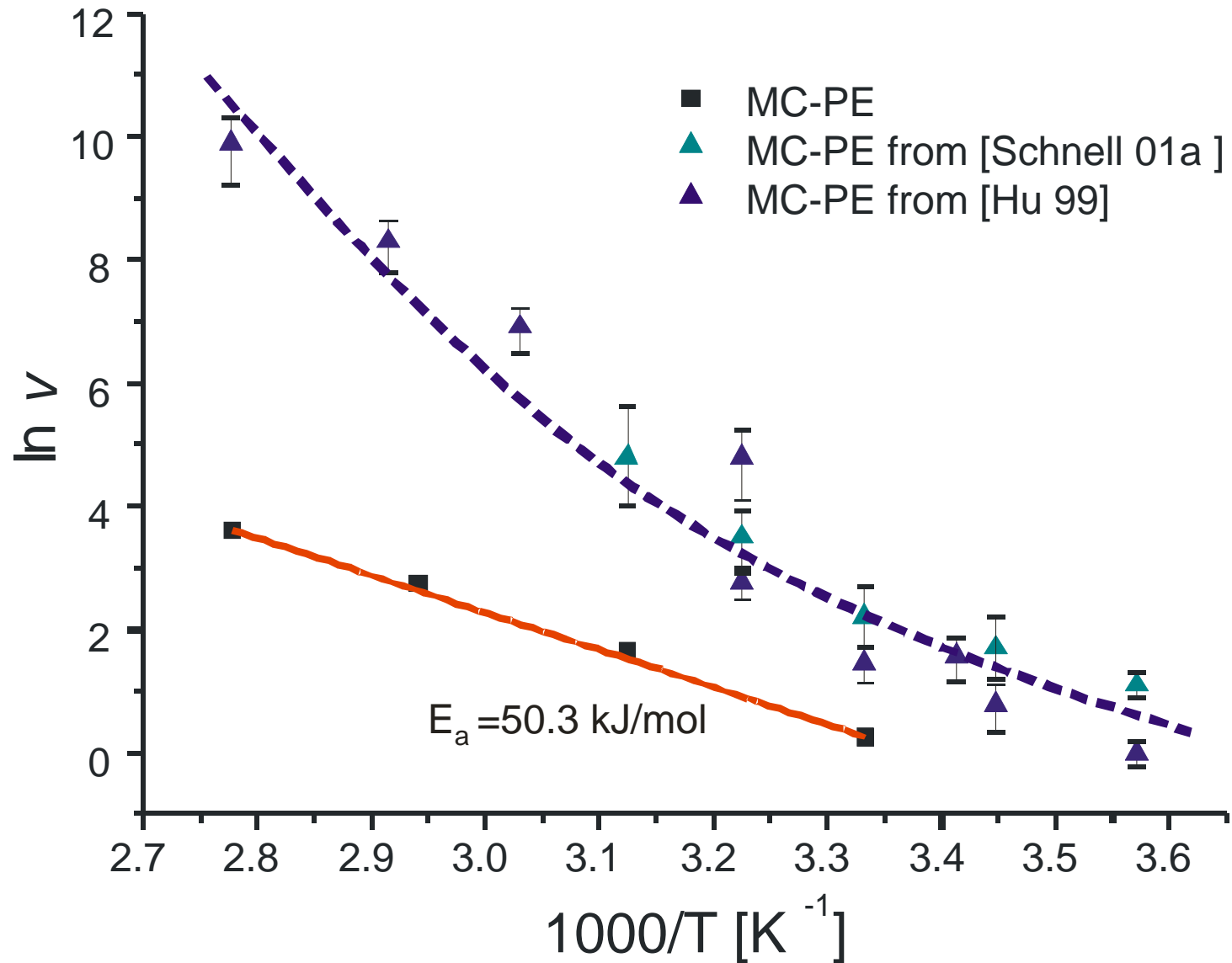
Chain Diffusion in Polyethylene



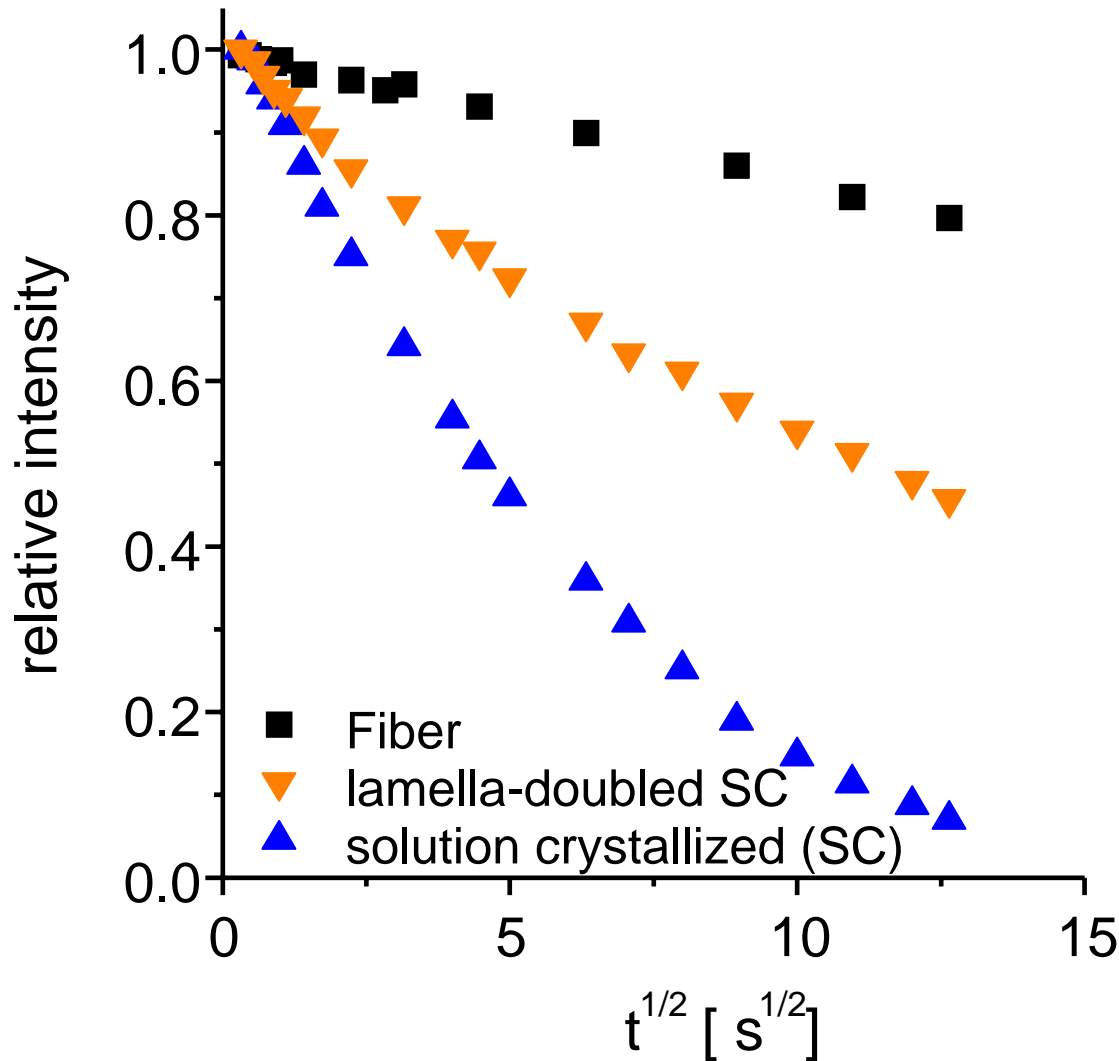
Activation Energy of Chain Motion



Chain Diffusion vs Local Jump Rate



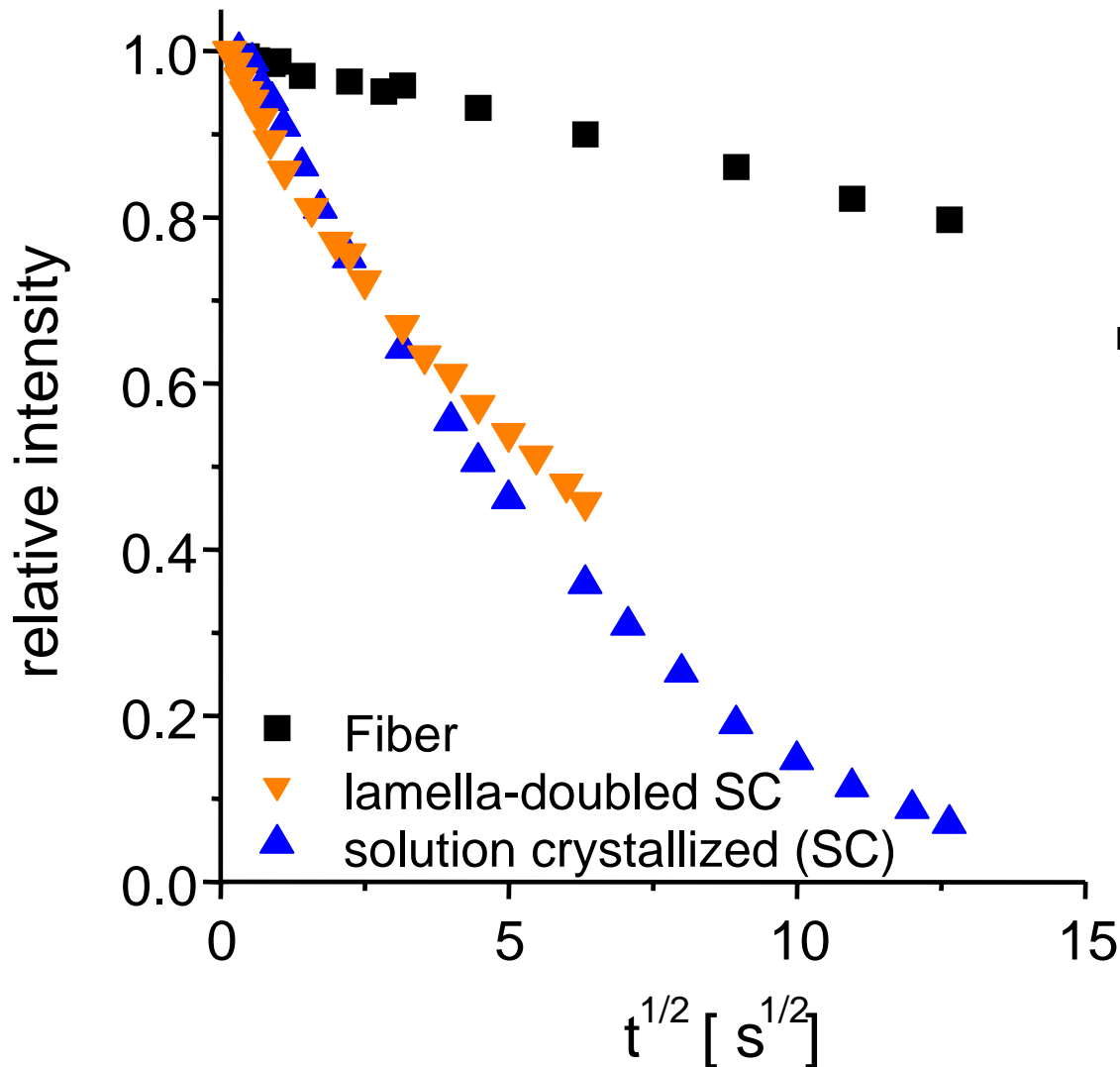
Variation of Lamellar Thickness



Annealing lamellar, solution crystallized polyethylene close to the melting point leads to lamellar doubling.

Rastogi et al, *Macromolecules* **30**, 7880 (1997).

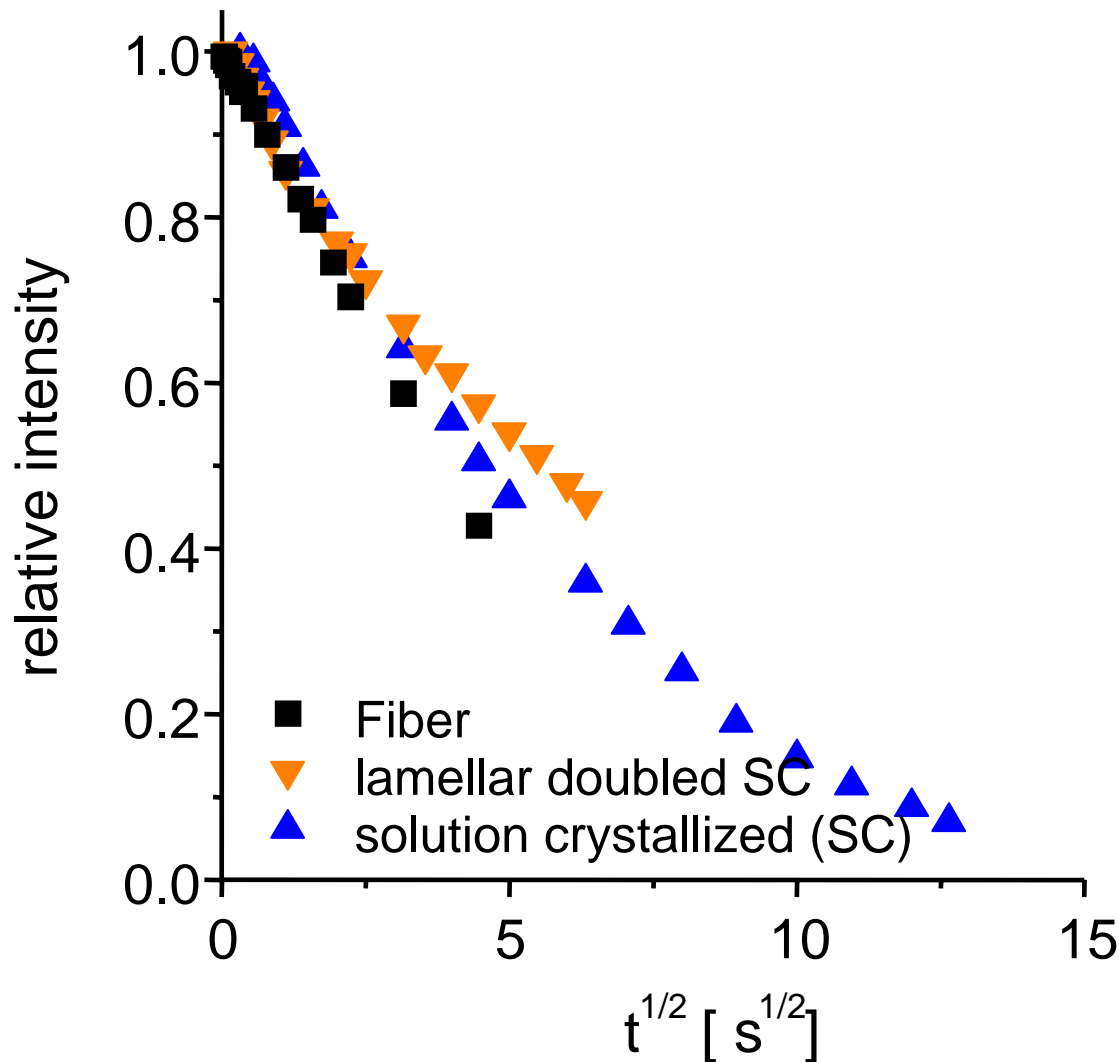
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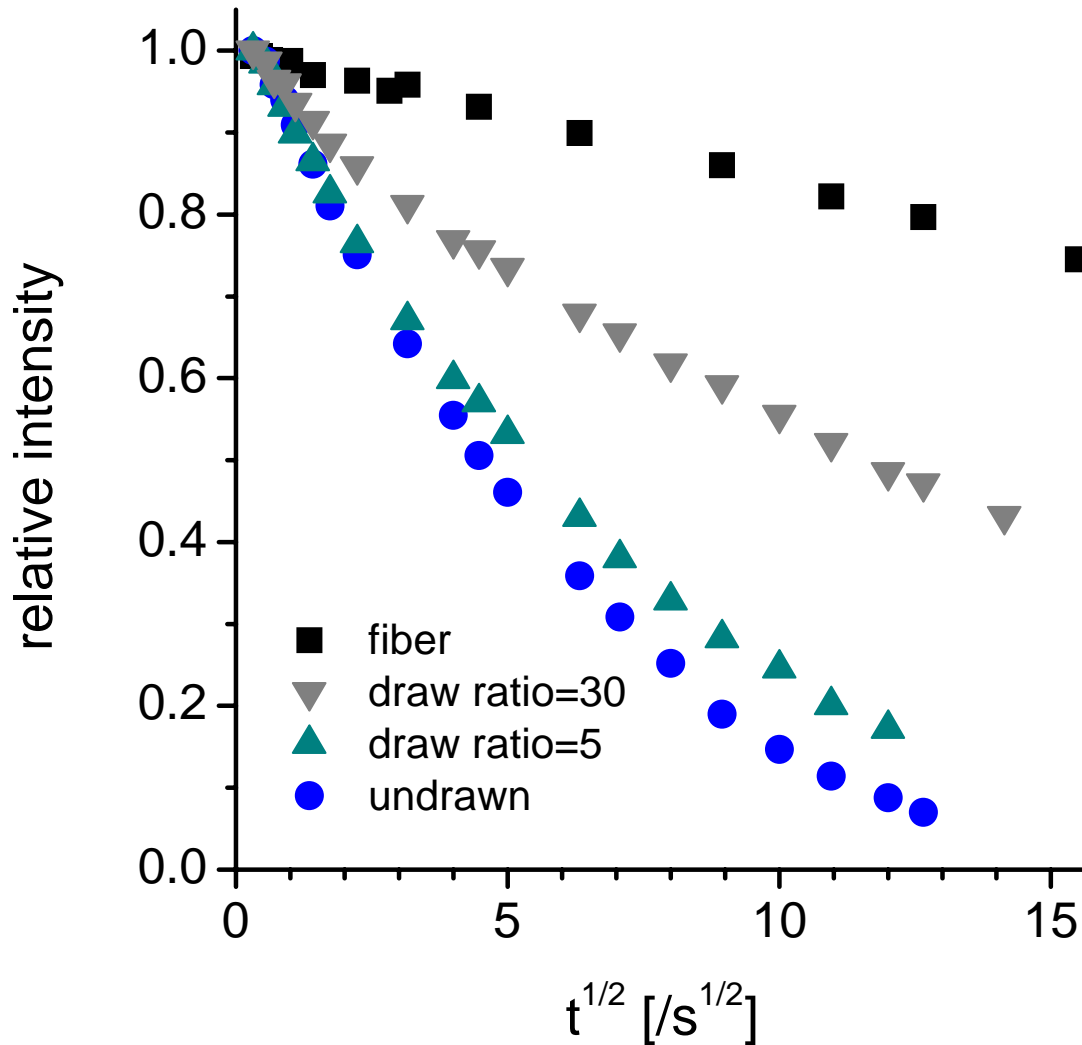
Rastogi et al, *Macromolecules* **30**, 7880 (1997).

time dependence of chain diffusion in the fiber sample

~ 100 nm

lamellar thickness.

Chain Diffusion in Drawn Samples



Weak deformations do not change the local morphology

Stronger deformations change the thickness of crystalline layers



Anisotropic NMR Interactions in SC-PE

- highly restricted dynamics
- almost temperature independent

Observation of Chain Translation

- local chain diffusion can be observed
- (NMR) crystallinity can be determined
- studies of local structure de/formation



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Prof. Dr. H.W. Spiess

University of Eindhoven

Prof. Dr. S. Rastogi

Dr. D.R. Lippits