

Response of Prestretched Nematic Elastomers to External Fields

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Outline

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- 2 Elasticity Including Nonlinear Relative Rotations
 - Energetics
 - Perpendicular Stretching
- 3 Linear Response under Pre-Strain
 - Effective Linear Shear Modulus
 - Director Reorientability
- 4 Conclusions



Monodomain Side-Chain Nematic Elastomers

Experiment:

- linear anisotropic elasticity
- nonlinear stress-strain plateau for perpendicular stretching
- accompanied by a complete director reorientation

Description and Interpretation:
effective linear modulus and director relaxation under pre-strain?



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Results

Stretching a mono-domain nematic elastomer perpendicularly
the resulting elastic plateau at finite strains

- comes with a vanishing effective linear modulus and a divergent director reorientability at its beginning and end (**soft mode**)
- this bifurcation-type behavior is a genuine manifestation of the role of **nonlinear relative rotations**
- it requires **two independent preferred directions** and discriminates nematic LSCEs from simple anisotropic solids

and

- this **soft mode behavior is not related** to the proposed **Nambu-Goldstone mode** ("soft-elasticity"), nor is any closeness to an ideal soft-elastic behavior ("semi-soft elasticity") required:
- the described scenario is found also for cases, where the plateau starts at very large applied strains



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Elastic and Orientational Degrees of Freedom

Network: $da_\alpha = R_{\alpha j} \Xi_{jk} dr_k$

Eulerian strain tensor

$$\begin{aligned} \varepsilon_{ik} &= \frac{1}{2} [\delta_{ik} - \Xi_{ij} \Xi_{ik}] \\ &= \frac{1}{2} [\delta_{ik} - (\partial a_\alpha / \partial r_k)(\partial a_\alpha / \partial r_i)] \\ &= \frac{1}{2} [\partial u_i / \partial r_k + \partial u_k / \partial r_i - (\partial u_j / \partial r_i)(\partial u_j / \partial r_k)] \end{aligned}$$

Nematic: Director

$$\hat{n} = S \cdot \hat{n}_0 \quad \text{and textures } (\nabla_j n_i)$$



Relative Rotations

Coupling:

- rotations of the **anisotropic network** $\hat{\mathbf{n}}^{nw} = \mathbf{R}^{-1} \cdot \hat{\mathbf{n}}_0^{nw}$
(there is no closed expression for \mathbf{R}^{-1} in terms of $\partial u_j / \partial r_i$)
- rotations of the **nematic director** $\hat{\mathbf{n}} = \mathbf{S} \cdot \hat{\mathbf{n}}_0$
- relative rotations** (projections)¹

$$\begin{aligned}\tilde{\mathbf{\Omega}} &\equiv \hat{\mathbf{n}} - \gamma \hat{\mathbf{n}}^{nw} \\ \tilde{\mathbf{\Omega}}^{nw} &\equiv -\hat{\mathbf{n}}^{nw} + \gamma \hat{\mathbf{n}}\end{aligned}$$

with $\gamma \equiv \hat{\mathbf{n}} \cdot \hat{\mathbf{n}}^{nw}$ resulting in $\tilde{\mathbf{\Omega}} \cdot \hat{\mathbf{n}}^{nw} = 0 = \tilde{\mathbf{\Omega}}^{nw} \cdot \hat{\mathbf{n}}$

¹A. M. Menzel, H. Pleiner and H. R. Brand, *J. Chem. Phys.* **126** (2007) 234901.



Free Energy

Power series expansion in ε_{ij} , $\tilde{\Omega}_i$, $\tilde{\Omega}_j^{nw}$, and n_i and all its couplings up to some order

here: simplified model (analytical treatment) - elastic nonlinearities neglected

$$\begin{aligned}
 F = & c_1 \varepsilon_{ij} \varepsilon_{ij} + \frac{1}{2} c_2 \varepsilon_{ii} \varepsilon_{jj} + \dots \\
 & + \frac{1}{2} D_1 \tilde{\Omega}_i \tilde{\Omega}_i + D_1^{(2)} (\tilde{\Omega}_i \tilde{\Omega}_i)^2 + D_1^{(3)} (\tilde{\Omega}_i \tilde{\Omega}_i)^3 \\
 & + D_2 n_i \varepsilon_{ij} \tilde{\Omega}_j + D_2^{nw} n_i^{nw} \varepsilon_{ij} \tilde{\Omega}_j^{nw} \\
 & + D_2^{(2)} n_i \varepsilon_{ij} \varepsilon_{jk} \tilde{\Omega}_k + D_2^{nw,(2)} n_i^{nw} \varepsilon_{ij} \varepsilon_{jk} \tilde{\Omega}_k^{nw} \\
 & - \frac{1}{2} \epsilon_a (n_i E_i)^2
 \end{aligned}$$

reduces in linear order to de Gennes' expression



Plateau for Perpendicular Stretch - Eulerian

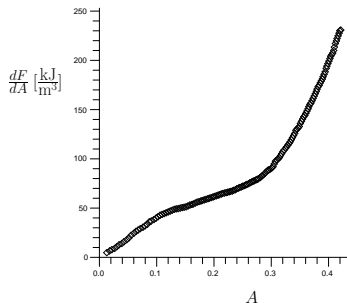


Fig.1: Stress-strain data measured by Urayama et al.^a transferred to the representation in terms of the stretch amplitude $A = \partial u_z / \partial z$ and dF/dA .

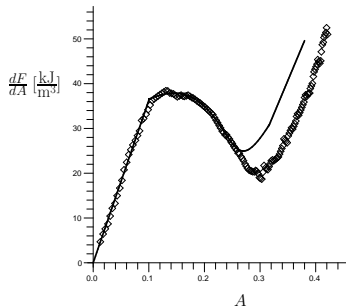


Fig.2: Same stress-strain data as in Fig.1 with nonlinear purely elastic contributions by the network of polymer backbones subtracted. The line is the result of the theoretical model^a

^aK. Urayama, R. Mashita, I. Kobayashi, and T. Takigawa, *Macromol.* **40** (2007) 7665.

^aA. Menzel, H.P., and H.R. Brand, *J. Appl. Phys.* **105** (2009) 013503.



Plateau for Perpendicular Stretch - Lagrangian

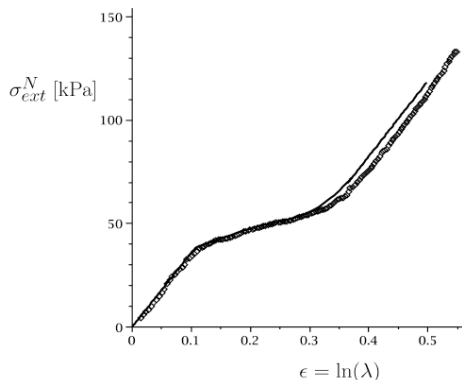


Fig.3: The same stress-strain data points of Urayama et al. and the theoretical line obtained by the present model (with the nonlinear elastic experimental contributions added) – now in the representation of the nominal stress as a function of the true strain.



Director Reorientation

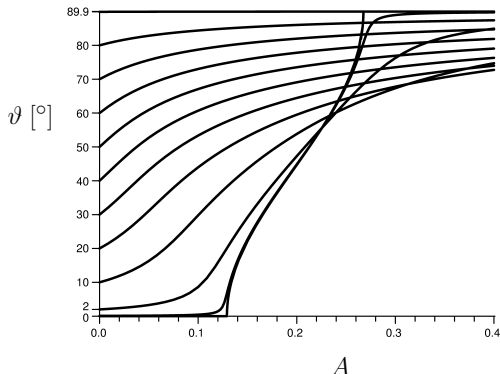


Fig.4: Angle ϑ between the director orientation and the x axis under the influence of an externally imposed strain A for various initial director orientations $\vartheta_0 = \vartheta(A = 0)$, e.g. 0° , 0.1° , 2° , 10° , \dots , 80° , and 89.9° , respectively. For $\vartheta_0 = 0^\circ$ (perpendicular stretch) a singular threshold behavior is found.



Forward bifurcation

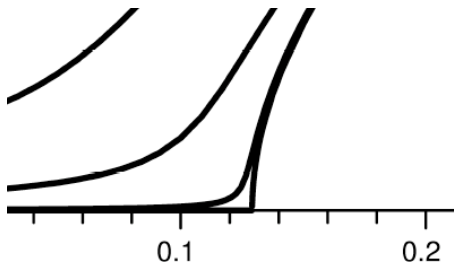


Fig.4a: $v = v(A)$; same as Fig.4 with the area around A_c enlarged

In the vicinity of A_c an **amplitude equation** can be derived analytically for the case $v_0 = 0$

$$0 = v \{ a(A_c - A) + g v^2 \} + \mathcal{O}(v^5).$$

→ forward bifurcation with exchange of stability between

$$v = 0 \text{ for } A < A_c \text{ and } v \sim \sqrt{A - A_c} \text{ for } A > A_c$$

for $v_0 > 0$ an imperfect bifurcation is obtained



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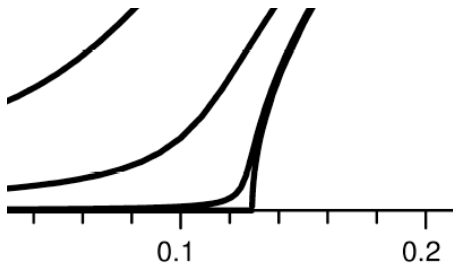


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for $v_0 > 0$ an **imperfect bifurcation** is obtained



Soft mode

- a **forward bifurcation** is similar to a **second order phase transition**
- an (**effective**) **susceptibility vanishes at the phase transition (at onset)**
- giving rise to diverging fluctuations (**soft mode**)
- in contrast to Nambu-Goldstone modes, where a susceptibility is identically zero throughout the whole phase due to symmetry reasons
- example: director rotations in a smectic C phase:
azimuthal (on the cone) Nambu-Goldstone mode
tilt angle: soft only at the smectic A to C transition
- for imperfect bifurcations no diverging fluctuations



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Homeotropic geometry

For a given prestrain A – that results in a given compression B , shear S , and tilt angle ϑ

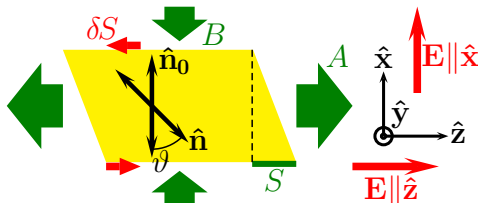


Fig.5: Homeotropic geometry

- 1 a small shear δS is added and the effective shear modulus is calculated
- 2 an external field is applied (\parallel and \perp to \hat{n}_0) and the reorientability of the director is calculated



Effective linear shear modulus

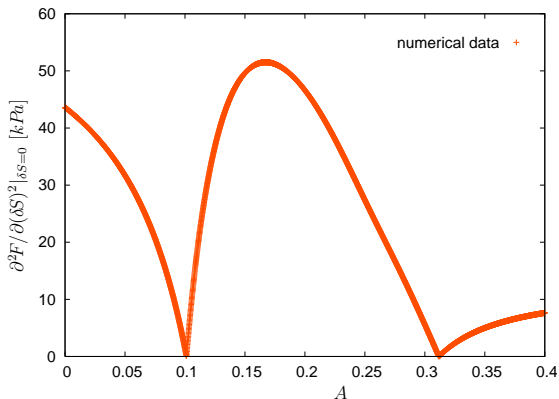


Fig.6: Effective shear modulus $\partial^2 F / \partial (\delta S)^2 |_{\delta S=0}$ as a function of the prestretching amplitude A . Here, the system is prestretched in a direction perfectly perpendicular to the initial director orientation $\hat{\mathbf{n}}_0$. The zeroes of the effective shear modulus at the beginning and end of the plateau denote diverging fluctuations.



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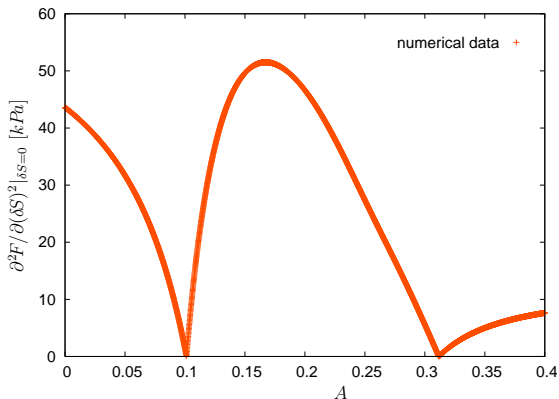


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Director reorientability

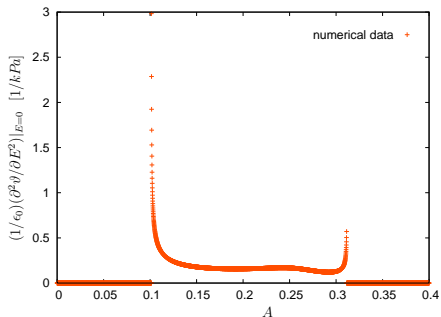


Fig.7: Reorientability $\partial^2 \vartheta / \partial E^2|_{E=0}$ as a function of the prestretching amplitude A , where the divergencies take place at the beginning and end of the plateau ($\mathbf{E} \perp \hat{\mathbf{n}}_0$)

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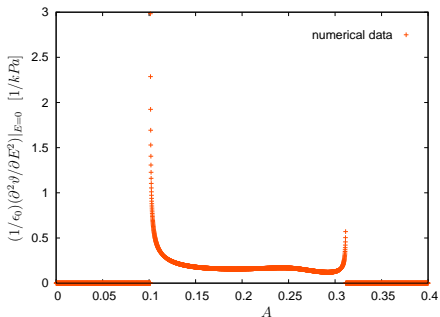


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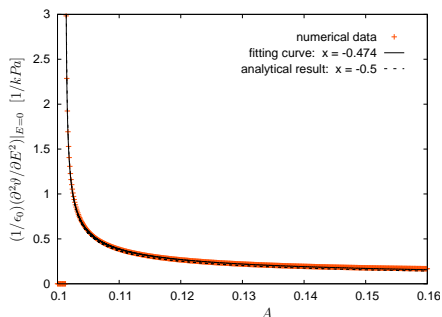


Fig.8: Same theoretical data fitted in the region $\vartheta \gtrsim 0$ by a curve $\propto (A - A_c)^x$ with $x \approx -1/2$, thus clearly indicating a soft mode behavior in mean field description



Oblique Pre-Strain

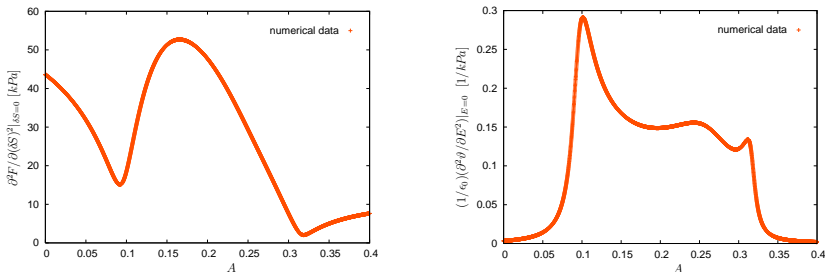


Fig.9: Effective shear modulus $\partial^2 F / \partial (\delta S)^2|_{\delta S=0}$ (left) and reorientability $\partial^2 \vartheta / \partial E^2|_{E=0}$ (right) as a function of the prestretching amplitude A . Here, the initial director orientation \hat{n}_0 slightly deviates from the perfectly perpendicular orientation by an angle of 0.01 rad (0.57°).



imperfect bifurcation: no divergent fluctuations²

² A. Petelin and M. Čopič, presentations at the *European Conference on Liquid Crystals*, Colmar, April 2009

Remarks

the fluctuations at the bifurcation **do not diverge** (the effective linear modulus remains non-zero)

- for an **oblique** prestretch
- due to **boundary** induced director **inhomogeneities** (necking)
- due to macroscopic **material inhomogeneities**
- if the **fluctuations** are treated **nonlinearly**³

there is no bifurcation (no diverging fluctuations)

- in the planar geometry (the small shear added is not in the director reorientation plane)
- for an external field in y direction (perpendicular to the director reorientation plane)

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Semisoftness

- the general scenario – elastic plateau with vanishing effective linear modulus at its beginning and end – has also been described by other methods⁴
- often, it is connected to semi-softness, where a small parameter α describes the (small) deviation from ideal softness;⁵

the plateau starts at $\lambda_1 \approx 1 + \alpha$ and the slope of the plateau is $3\mu\alpha$ (cf. Chaps. 7.4 and 7.5 of Ref. 5)

- however, the smallness of $A_c \approx 0.1$ (corresponding to $\alpha \approx 0.1$) is not a necessary condition for the soft mode behavior at the beginning and end of the plateau

⁴ J. S. Biggins, E. M. Terentjev, and M. Warner, *Phys. Rev. E* **78** (2008) 041704
and F. F. Ye and T. C. Lubensky, *J. Phys. Chem. B* **113** (2009) 3853

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High Plateau

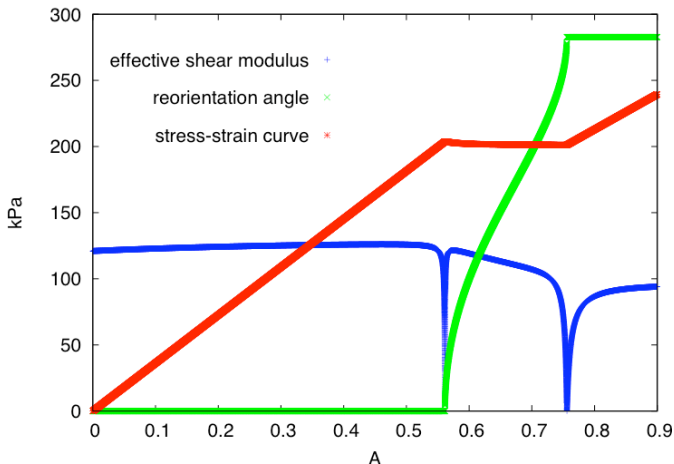


Fig.10: In this case the plateau starts at a rather **large pre-strain** $A_c \approx 0.56$ (or $\lambda \approx 2.3$) and ends at $A \approx 0.76$ (or $\lambda \approx 4.2$)
 – the scenario is the same as for very small A_c .



Summary

- the scenario of an elastic plateau at finite perpendicular stretching, with a vanishing effective linear modulus and a divergent director reorientability at its beginning and end (soft mode), is a genuine manifestation of an instability due to nonlinear relative rotations;
- it requires two independent preferred directions and discriminates these systems from simple anisotropic solids;
- there is no need for a small parameter⁶, nor for the closeness to an ideal soft-elastic behavior (Nambu-Goldstone or almost Nambu-Goldstone mode)
 - the soft mode scenario can happen, even when the plateau starts at very high strains

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Lagrange description

Comparing the initial dimension l_0 to the actual dimension (in the direction of the external force F_{ext}), the ratio

$$\lambda = \frac{l}{l_0} \quad (1)$$

is taken as a measure of the induced strain.

Sometimes, the so called true strain $\epsilon = \ln(\lambda)$ is taken as a variable.

Stresses are recorded either as true stress

$$\sigma_{ext} = \frac{F_{ext}}{l_x l_y} \quad (2)$$

or as nominal stress

$$\sigma_{ext}^N = \frac{F_{ext}}{l_{x,0} l_{y,0}}, \quad (3)$$

From the experimental point of view the initial dimension l_0 is considered to be constant and the current sample dimension l is changed.



Eulerian description

In the hydrodynamic (Eulerian) picture the current dimension of the sample l is considered to be constant, and what changes is the initial dimension l_0 . For the displacement field $u_z = Az$ (or $l_z - l_{z,0} = Al_z$) the strain is

$$\lambda = \frac{1}{1 - A}. \quad (4)$$

and the stresses are

$$\sigma_{ext} \equiv \frac{F_{ext}}{l_x l_y} = \frac{dF}{dA}, \quad (5)$$

$$\sigma_{ext}^N \equiv \frac{F_{ext}}{l_{x,0} l_{y,0}} = (1 - A) \frac{dF}{dA}. \quad (6)$$

Here, the expressions on the left of Eqs. (5) and (6) are given as functions of λ , the expressions on the right as functions of A .

The connection between both follows from Eq. (4).



Constrained equilibrium

As an ansatz we use for the displacement fields

$$u_z = Az + Sx, \quad (7)$$

$$u_x = Bx \quad (8)$$

$$u_y = Cy. \quad (9)$$

and for the director orientation

$$\hat{\mathbf{n}} = (\cos \vartheta, 0, \sin \vartheta) \quad (10)$$

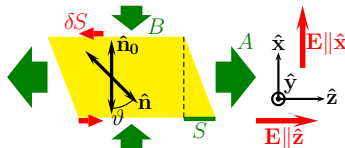


Fig.5: homeotropic geometry

For a given initial orientation ϑ_0 and external stretch A , the values $S(A)$, $B(A)$, and $\vartheta(A)$ follow from the equilibrium conditions $\partial F / \partial S = 0$, $\partial F / \partial B = 0$, and $\partial F / \partial \vartheta = 0$ (the compression C follows from the incompressibility condition).



Shear and compression

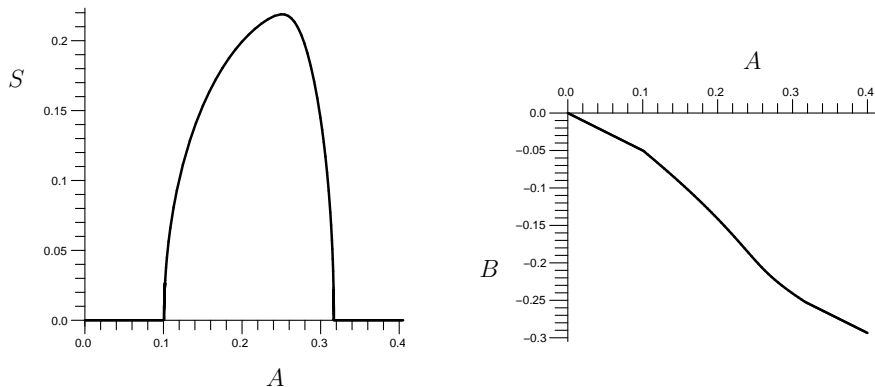


Fig.11: The shear $S(A)$ and the compression $B(A)$ as a function of the pre-strain amplitude A .

Effective linear modulus

For each given pre-strain A a small shear δS is added,

$$u_z = Az + [S(A) + \delta S]x \quad (11)$$

and the free energy (including δS) is again minimized w.r.t. ϑ and then calculated to lowest order in δS

$$F_A = \frac{1}{2}c_{\text{eff}}(A)(\delta S)^2 + \mathcal{O}(\delta S)^3 \quad (12)$$

The effective linear modulus $c_{\text{eff}}(A) = \partial^2 F_A / \partial (\delta S)^2 |_{\delta S=0}$ is shown in Figs. 6 and 9.

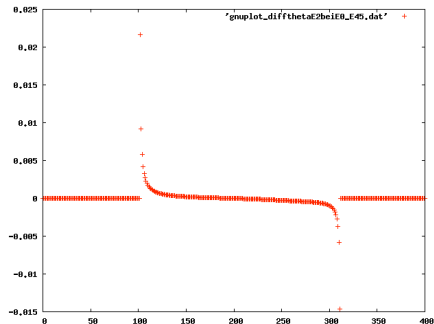


Orientability

For given external field \mathbf{E} in the x-z plane and a given pre-strain A , the system of constrained equilibrium conditions $\partial F/\partial\vartheta = \partial F/\partial B = \partial F/\partial S = 0$ is solved, resulting in $F = F(\mathbf{E}, A)$.

For each value of A , there is $\partial\vartheta/\partial E|_{E=0} = 0$ due to stability reasons.

Therefore, we take the second derivative $\partial^2\vartheta/\partial E^2|_{E=0}$ as a measure for the reorientability of the director $\hat{\mathbf{n}}$ in an external field for a given stretching amplitude A . In Fig.7 this reorientability is shown for $\mathbf{E} \perp \hat{\mathbf{n}}_0$, while for $\mathbf{E} \parallel \hat{\mathbf{n}}_0$ the sign of it is reversed. The case $E_x = E_z$ is shown on the right



Divergence of the orientability

- the coefficients of the amplitude equation close to the threshold ($E^2 = E_{x,z}^2$)

$$0 = \vartheta \{ a(A_c - A) + g\vartheta^2 \} + \mathcal{O}(\vartheta^5). \quad (13)$$

generally acquire field contributions $\sim E^2$ due to the dielectric anisotropy energy

- in the limit $E \rightarrow 0$ one can write, e.g. $A_c(E) = A_c(1 + \zeta_A E^2)$
- for $A \gtrsim A_c$ this leads to the field dependence of the tilt angle

$$\vartheta = \sqrt{\frac{a}{g(E)}(A - A_c(E))} \approx \sqrt{\frac{a}{g}(A - A_c)} \left(1 + \zeta E^2 + \frac{\zeta_A}{2} \frac{A_c}{A_c - A} E^2 \right) \quad (14)$$

- and to the orientability

$$\partial^2 \vartheta / \partial E^2 |_{E=0} \sim (A - A_c)^{-1/2} + \mathcal{O}((A - A_c)^{1/2})$$

which is observed in Fig.8.

